#### RILEM TC REPORT



# Test methods for chloride diffusivity of blended cement pastes: a review by RILEM TC 298-EBD

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**Abstract** The use of supplementary cementitious materials (SCM) is an important part of the roadmap for reducing CO<sub>2</sub> emissions and extending the service life of reinforced concrete structures. To accelerate the adoption of SCMs, the RILEM Technical Committee 298-EBD evaluates scaled-down cement paste test methods to assess the effect of SCM on resistance to chloride and sulfate ingress and reactivity, which are critical to concrete durability. This review focuses on methods for measuring chloride diffusivity

and is divided into four sections: diffusivity models and parameters, diffusion test methods (including NMR and chloride measurements), migration test methods and implications for future research. Key insights highlight the complexities of multi-species ionic and molecular diffusion/migration, including various binding interactions, and compares the different measurement methodologies. The review also addresses the test scale and aggregate effects, noting the pros and cons of testing at the paste, mortar, and

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324 Page 2 of 35 Materials and Structures (2025) 58:324

| concrete scales. The review underscores the need for further investigation into testing protocols and the                   |  |  | EDS                     | Energy dispersive X-ray spectroscopy |
|---|--|--|-------------------------|--------------------------------------|
|   | · ·  | oride diffusion, emphasizing   | EPMA                    | Electron probe                       |
|   |  |  | Livin                   | micro-analyser                       |
|   |  |  | FA                      | Fly ash                              |
| that comprehensive testing across different scales provides complementary information for assessing durability performance. |  |  | GGBFS                   | Ground granulated blast-             |
| bility performance.   |  |  | GGDIS                   | furnace slag                         |
|   | <b>Keywords</b> Chloride ing                                     | race . Diffusion tasts .   | НТО                     | Tritiated water                      |
|   | Migration test · Cement pa                                       |  | ICP                     | Inductively-coupled                  |
|   | Supplementary cementition  |  | ici                     | plasma                               |
|   | Supplementary cementing  | ous materials (SCIVI)  | LIBS                    | Laser-induced breakdown              |
|   | Abbreviations  |  | LIDS                    | spectroscopy                         |
|   | Shorthand notations µX   | RF Micro X-ray   | LA                      | Laser ablation                       |
|   | Shorthand notations µ2.  | fluorescence   | LL                      | Limestone filler                     |
|   |  | spectroscopy   | PC                      | Portland cement (OPC –               |
|   | AFm  | Alumino-Ferrite mono-  | FC                      | · ·                                  |
|   | Al'III   | substituted calcium  | MS                      | ordinary portland cement)            |
|   |  |  | PIXE                    | Mass spectrometry                    |
|   |  | hydrate phases (general  | PIXE                    | Particle-induced X-ray               |
|   |  | formula [Ca <sub>2</sub> (Al,Fe)   | DI C                    | emission                             |
|   |  | $(OH)_6$ ]·X·yH <sub>2</sub> O, where X  | PLC                     | Portland cement blended              |
|   |  | is OH <sup>-</sup> , ½SO <sub>4</sub> <sup>2-</sup> , ½CO <sub>3</sub> <sup>2-</sup> , | DCM                     | with limestone                       |
|   | A 174  | Cl <sup>-</sup> )  | RCM                     | Rapid chloride migration             |
|   | AFt  | Alumino-Ferrite tri-   | RT                      | Reactive transport                   |
|   |  | sulfate calcium hydrate  | SCM                     | Supplementary cementi-               |
|   | D.4  | (Ettringite)   | 99.4                    | tious materials                      |
|   | BA   | Biomass ash  | SSA                     | Sewage sludge ash                    |
|   | CB   | Crushed brick  | TD                      | Thermodynamics (Chemi-               |
|   | CBI  | Chloride binding isotherm  |                         | cal equilibrium)                     |
|   | CC   | (conventional on powders)  | TXM                     | Transmission X-ray                   |
|   | CC   | Calcined clay  |                         | microscopy                           |
|   | EDL  | Electrical double layer  | Mathematical parameters | $C_{\rm S}$ Total chloride           |
|   |  |  |                         | concentration at                     |
|   |  |  |                         | the specimen sur-                    |
|   | F. Georget<br>Institute of Building Materials                    | Pasagrah PWTH Agahan   |                         | face (exposed to                     |
|   | University, Aachen, Germany                                      | Research, RW 111 Machen  | ~                       | chloride solution)                   |
|   | , , , , , , , , , , , , , , , , , , ,                            |  | $C_{\mathrm{d}}$        | Total chloride concentra-            |
|   | M. Mrak · S. Dolenec   |  |                         | tion at penetration depth            |
|   | Slovenian National Building an<br>Institute, Ljubljana, Slovenia | nd Civil Engineering   |                         | $x_d$ where colour changes in        |
|   | mstitute, Ljubijana, Slovema                                     |  | _                       | a colorimetric method                |
|   | S. Dolenec   |  | D                       | A diffusion coefficient in           |
|   | Department of Geology, Facul                                     |  | _                       | general terms                        |
|   | and Engineering, University of Slovenia                          | ł Ljubljana, Ljubljana,  | $D_0$                   | Self-diffusion coefficient           |
|   | Siovenia   |  |                         | (tabulated values for                |
|   | T. Völker  |  |                         | infinite dilution in water)          |
|   | Federal Institute for Materials                                  | Research and Testing,  |                         | related to random molecu-            |
|   | Berlin, Germany  |  |                         | lar motion in the absence            |
|   | P. Suraneni  |  |                         | of a concentration gradient          |
|   |  |  |                         |                                      |



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Materials and Structures (2025) 58:324 Page 3 of 35 324

| D              | Apparent diffusion coef-                       |
|----------------|--|
| $D_{app}$      | ficient (vague usage,                          |
|                | typically referring to                         |
|                | combined chemical and                          |
|                | physical nature)                               |
| D              | A general diffusion coef-                      |
| $D_{ m F1}$    | ficient used in Fick's first                   |
|                | law  |
| D              | Effective diffusion coef-                      |
| $D_{\it eff}$  |  |
|                | ficient (vague usage, typically referring only |
|                |  |
|                | to theoretical physical                        |
|                | morphological nature)                          |
|                | driven by concentration or                     |
| D              | activity gradients                             |
| $D_{ m intra}$ | Intraphase diffusion coeffi-                   |
|                | cient of a species within a                    |
|                | specific phase (e.g., water,                   |
|                | solution, pore solution)                       |
|                | driven purely thermally, or                    |
|                | by specific gradients                          |
| $D_{ m inter}$ | Interdiffusion (or mutual                      |
|                | diffusion) coefficient due                     |
|                | to a net movement of                           |
|                | multiple species driven by                     |
| _              | concentration gradients                        |
| $D_{ m ssm}$   | Steady-state migration dif-                    |
| _              | fusion coefficient                             |
| $D_{ m ssd}$   | Steady-state diffusion                         |
| _              | coefficient                                    |
| $D_{ m nssd}$  | Non-steady-state diffusion                     |
|                | coefficient                                    |
| $D_{ m nssm}$  | Non-steady-state migra-                        |
| _              | tion diffusion coefficient                     |
| P              | Porosity                                       |
| w/b            | Water-to-binder mass ratio                     |
| w/c            | Water-to-cement mass                           |
|                | ratio  |
| $x_d$          | Penetration depth where                        |
|                | colour changes in a colori-                    |
|                | metric method                                  |

#### 1 Introduction

The use of supplementary cementitious materials (SCM) can reduce CO<sub>2</sub> emissions by partially replacing cement and extending the service life of reinforced concrete structures. For durability [1],

it is important to understand how SCMs influence long-term chloride ingress in cementitious materials [2], a focus of the RILEM Technical Committee 298-EBD. The committee evaluates down-scaled cement paste test methods to assess SCM effects on resistance to chloride and sulfate ingress which significantly impact concrete structures durability. The committee is divided into four sub-working groups: WG1 – Microstructure, WG2 – Reactivity with chlorides, WG3 - Reactivity with sulfates, and WG4 - Transport with and without reactivity. WG4 aims to evaluate cement paste test methods to quantify the effects of new SCM on transport-diffusivity properties related to chloride (and sulfate) ingress. The term 'reactivity' refers to two aspects: 1) chloride binding reactions and 2) ongoing hydration of cement and SCM. 'Without reactivity' pertains to purely diffusive mechanisms in inert porous media, excluding concurrent binding and hydration. To study chloride ingress in concrete, standard methods like bulk diffusion or accelerated migration tests have been developed. This review concentrates on various experimental methods to obtain a diffusion coefficient D derived from chloride (flux or profile) measurements (Table 1). Chloride flux measurements are relevant for steady state through diffusion tests, typically measuring changes in the exposure solution, while non-steady-state methods involve solid profile measurements. Shi et al. [2] reviewed existing methods, discussing pros and cons. Recent work (e.g. [3]) showed that scaling diffusivity tests to cement paste reduces variables, labor and testing time. The overall goal of the subgroup 4 is on evaluating these scaled-down methods to assess SCM effects on cement paste's resistance against chloride ingress.

This review paper is organized into four main sections to systematically address chloride diffusivity in cementitious materials.

# 2 Chloride diffusivity models and key parameters

To avoid confusion, it is crucial to clearly define various forms of diffusion coefficients (D's) using the models and/or experiments utilized to derive them. Chloride ingress in cementitious materials can be mathematically described using modelling approaches of different complexities. The service life extrapolation capability is strongly affected by the



324 Page 4 of 35 Materials and Structures (2025) 58:324

**Table 1** Common standard methods to measure ionic diffusivity in concrete, along with related sections in the structure of the current paper, the measured parameters and selected references to standards and/or reference papers

| Category  | Short description               | Section | Parameter                                       | Standards (or reference papers)   |
|-----------|---------------------------------|---------|---|---|
| Diffusion | Self-diffusion                  | 3.1     | $D_{0,} D_{\rm intra}$                          | [4, 5]  |
|           | Steady state through diffusion  | 3.2     | $D_{\mathrm{ssd}}\left(D_{\mathrm{eff}}\right)$ | None [6]  |
|           | Non-steady-state bulk diffusion | 3.3     | $D_{\rm nssd}  (D_{\rm app})$                   | ASTM C1556 [7], NT build 443 [8], ISO 1920–11 [9], EN 12390–11 [10]; Chloride profiles by NT Build 208 [11] or more detailed RILEM recommendations [12, 13] |
| Migration | Steady state                    | 4.1     | $D_{ m ssm}$                                    | NT Build 355 [14], XP P 18–461 [15]   |
|           | Non-steady-state (RCM)          | 4.2     | $D_{ m nssm}$                                   | NT BUILD 492 [16], French XP P 18–462 [17], EN 12390–18 [18], Spanish UNE 83987 [19] and UNE 83992–2 [20]   |

level of model simplification. Simple chloride (and carbonation) ingress models are empirical single-species analytical solutions of diffusion equations mainly for curve-fitting [21–23]. More complex models [24], such as reactive transport multi-species approaches [25–27] require a reliable diffusion coefficient ( $D_{\rm eff}$ ), chloride binding and other reaction mechanisms as key input parameters.

Scaling laboratory D values to field conditions remains a key challenge in predicting concrete's longterm durability. The accelerated lab experimental results, typically obtained in weeks (or days), have limited value to extrapolate the service-life of concrete structures [22, 28]. Thus, a better understanding of the combination between modeling and various lab experimental methods available to obtain D parameters is of critical importance. Extrapolations based on scientific principles and calibrations using such (accelerated) datasets, could significantly enhance the accuracy of service-life predictions. Here, complications due to missing data and methodologies on how to evaluate the effects of SCM on the chloride ingress, makes the search for adequate models and input parameters even more relevant.

### 2.1 Parameters influencing chloride diffusivity

Chloride ingress in concrete involves multiple interacting factors, including concrete-scale effects, chloride binding, porosity, pore solution and environmental conditions (Fig. 1). Decoupling these parameters is essential for understanding mechanisms and enabling long-term performance prediction. Without Cl binding, the diffusion of substances in hydrated cement paste is primarily controlled by the interconnected capillary pores, influenced by factors like

water/binder ratio, binder (degree of) hydration, and material properties. Pozzolanic and hydraulic SCMs enhance binder pore connectivity and tortuosity (including pore constrictivity), reducing diffusivity. However, inadequate curing, especially with SCM and low water/binder ratios, can lead to insufficient hydration near surfaces, increasing D. Air voids, porous aggregates, and aggregate-paste interfaces also impact diffusion pathways. Effects of hydration reactivity, chloride binding, binder and pore solutions chemistries are discussed by Weerdt et al. [23]. The influence of both material-related and environmental factors on chloride diffusivity in concrete is summarized in Table 2, highlighting key parameters such as water-to-binder ratio, SCM use, pore structure, temperature, and relative humidity.

# 2.2 Empirical and more deterministic model approach

# 2.2.1 Model based on $D_{app}$ ( $D_{nssd}$ , $D_{nssm}$ )

Shafikhani et al. [29] reviewed analytical (mostly empirical) mathematical models for concrete chloride diffusion, based on tests for  $D_{\rm nssd}$  and  $D_{\rm nssm}$ , analyzing a wide range of experimental data, including SCM effects (silica fume, fly ash, slags, limestone) with various cement replacement dosages, as well as the aggregate volume fraction, cement content and max aggregate size. Results show that models incorporating capillary porosity, pore structure, diffusivities, aggregate volume, and ITZ properties deliver the highest accuracy and precision. Some models have used  $D_{\rm app}$  derived from bulk diffusion tests without explicitly accounting for chloride binding [7], while others used  $D_{\rm eff}$ , incorporating chloride binding



Materials and Structures (2025) 58:324 Page 5 of 35 324

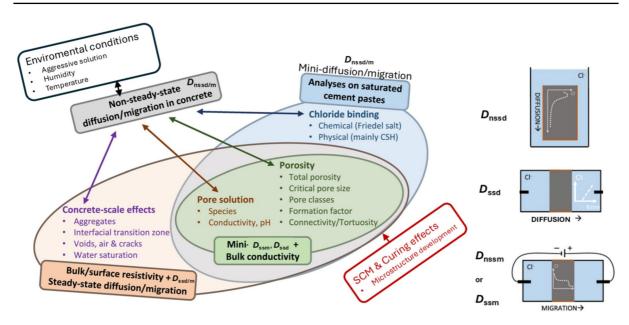


Fig. 1 Key parameters of chloride ingress included in the different types of experimental tests at different scales

Table 2 Key parameters affecting chloride diffusivity in concrete

| Parameter type | Category                                   | Specific factor   | Rationale/impact   |
|----------------|--|---|--|
| Material       | Water-to-binder ratio                      | Low w/b ratio   | Reduces pore connectivity; lowers diffusivity                |
|                | SCMs (e.g., slag, fly ash, calcined clays) | Alters binding (↑AFm), lowers pH, refines pores           | Enhances chloride binding; affects pore solution composition |
|                | Clinker dilution                           | Less C-S-H formed   | Reduces binding capacity                                     |
|                | Aggregate design                           | Optimized content and grading                             | Reduces tortuosity, improves barrier effect                  |
|                | Air voids                                  | Controlled air content                                    | Saturation-sensitive voids influence transport               |
|                | Cracking                                   | Minimized cracking  | Cracks create fast paths for chloride ingress                |
|                | Pore morphology                            | SCMs refine pore size distribution                        | Smaller, disconnected pores hinder diffusion                 |
|                | Chloride binding                           | Strong chemical (AFm) & weak physical (C-(A-)S-H) binding | Buffers free Cl <sup>-</sup> ; impacts total transport       |
|                | Donnan exclusion                           | Ion exclusion in small pores                              | Lowers effective diffusivity                                 |
| Environmental  | Exposure solution                          | Use realistic ions (e.g., seawater, CaCl <sub>2</sub> )   | Affects binding chemistry and transport                      |
|                | Temperature                                | Control ambient/curing temperature                        | $\uparrow T$ significantly increases diffusion rate          |
|                | Leaching effects                           | NaCl diffusion, OH buffering                              | Alters pore chemistry, pH, and ionic strength                |
|                | Relative humidity                          | Partial saturation (RH < 100%)                            | Reduces continuity of water phase; lowers mobility           |

corrections, typically based on migration test data. Combined analytical and numerical multi-scale modeling provide a novel perspective to study the chloride penetration in concrete by considering the effects of both the multi-scale microstructural characteristics and multi-species interactions [29].

The commonly used  $D_{\rm app}$  (Fick's 2nd) approach Weerdt (2023) [23] is an empirical parameter

obtained by fitting concentration profiles to simplified (diffusion-like) models that lump together the diffusion mechanisms with mechanisms not explicitly included in the model [21, 22]. This engineering approach, used in many studies on site after long exposure to sea water [30, 31], reduces model complexity and the number of parameters, it may significantly limit the model's extrapolation



capabilities, especially under nonlinear concentration dependencies.

# 2.2.2 Model based on $D_0$ or $D_{eff}$

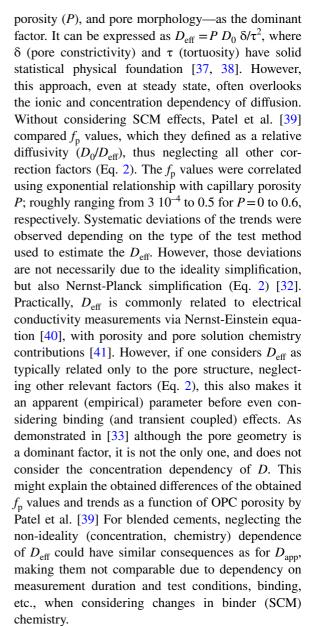
Tang [32] proposed a simple analytical model to describe the main electro-chemo-physical processes of chloride diffusion in cementitious materials. It is based on the binary (single salt) electrolytic solution diffusion theory. This laid a theoretical background for the chloride ion diffusion and migration processes, linking a pure electrolyte solution theory, from diluted to concentrated systems and extending it to concrete. In this paper, the use of mathematical expressions is limited to a bare minimum, by adapting Fick's first law using appropriate correction factors [33]. This approach adapts simple models, valid for ideal cases (e.g. Fick's diffusion), to more realistic scenarios by introducing physically meaningful correction factors (e.g. for activity dependencies). The  $D_{\rm F1}$  used in Fick's first law (Eq. 1), could be related to the flux (J) in concentrated pore solution defined by the gradient of chemical potential (or concentration c in ideal case) and  $D_0$  (an ideal reference case for selfdiffusion) through a set of coefficients (Eq. 2) [34]:

$$J = -D_{F1} \nabla \mu \cong -F_f D_0 \nabla c \tag{1}$$

$$F_f = f_{\gamma} f_E f_f f_p \tag{2}$$

The overall coefficient  $F_f$  should (separately) account for the different chemo-physical phenomena responsible for the non-linear dependence of the  $D_{FI}$ on the concentration (composition) of the solution ( $f_f$  $f_{\gamma} f_{\text{E.solution}}$ ) including surface charge effects ( $f_{\text{E.surface}}$ ) as well as the geometrical pore structure  $(f_p)$ . Briefly,  $f_{\gamma}$  accounts for activity non-ideality,  $f_{\rm E}$  for the electrical coupling (e.g. as a simplification of the Nernst-Planck equation [35] and effects of surface charges) that result in relative anion/cation mobility, and  $f_f$ accounts for friction effects in concentrated electrolytes. To address solid surface charge (Donnan) effects on electrolyte diffusion, a well-established analytical steady-state theory by Revil [36] could be employed, while transient models are needed to capture coupled dynamic processes including chloride binding.

In common terminology,  $D_{\rm eff}$  emphasizes pore geometry—related to  $D_0$  (ideal self-diffusion), open



Concentration dependence of  $D_{\rm eff}$  can be described using the intrinsic friction parameter  $\psi$ , as per Tang's model. This model has been recently refined [33] by calibrating the pore geometry factor ( $f_{\rm p}$ , Eq. 2) using same  $D_{\rm ssd}$  measurement data for OPC. A novel method to estimate  $\psi$  in OPC materials accounts for better separation of parameters, showing  $\psi$  to be 1.9 times higher than in pure NaCl solutions, significantly lower than Tang's values. This study systematically examines factors affecting  $D_{\rm ssd}$  chloride concentration dependency, highlighting  $D_{\rm ssd}$  test as a promising but time-intensive to understanding SCM effects.



Materials and Structures (2025) 58:324 Page 7 of 35 324

Advanced (multi-component) reactive-transport models present a significant challenge as various mechanisms are interlinked, making it highly challenging, if not impossible, to independently obtain the required model parameters. For instance, the coupling of Cl binding with D through multi-ion and surface-EDL interaction models leads to the emergence of more fundamental (intrinsic) D. The proposed simplified relations (e.g.,  $D_{\text{nssd}}$  vs.  $D_{\text{ssd}}$ ) operate under the assumption that certain effects, such as chloride binding, can be eliminated in steady-state diffusion tests. Performing tests with multiple upstream concentrations facilitates the determination of both the intrinsic  $D_{\rm eff}$  (though still away from its oversimplistic physical meaning), which only better approaches the true effective diffusivity coefficient at zero concentration extrapolation, and the concentration dependency of  $D_{\rm ssd}$  in relation to Cl-salts.

# 2.2.3 Relationships between diffusion coefficients

Many simplified relationships between the various D's are proposed in literature, and their limitations should be made clearer.  $D_{\text{nssd}}(D_{\text{app}})$  is typically (over) simplified by relating it to the  $D_{\rm ssd}$  ( $D_{\rm eff}$ ), porosity and chloride binding isotherm: e.g.  $D_{\text{nssd}} = D_{\text{ssd}}$  $(P+dC_b/dC_f)$ . Note that the binding term is constant only if the isotherm is linear [21, 42]. Analytical relations between the diffusion coefficients were established by Tang [32], based on 1:1 electrolyte theory of diffusion. For example, the link between  $D_{\text{ssd}}$  and  $D_{\rm ssm}$  depends on the Cl binding, concentration ( $dD_{\rm ssm}$ /dc) and potential gradient. However, how to consider such parameters quantitatively remains a key challenge. If neglecting binding in the (short transient) migration test, the following relation is suggested  $D_{\text{ssm}} = P D_{\text{nssm}}$  [43], however only as a crude first approximation as binding and other effects in migration are still unclear.

# 2.2.4 Effect of aggregates

Several studies in the literature show that beyond 50% by mass of aggregates a continuous pathway is formed between the ITZs of each aggregate (Winslow et al., 1994) and that most of the cement paste is ITZ [44, 45]. Although controversial today for concrete, this conclusion seems appropriate for mortars for which the quantity of aggregates is significant in

volume percentage. By neglecting the water adsorption capacities of the aggregates (assuming the aggregates are saturated with water) and by assuming a negligible diffusion rate in these same aggregates, Soive et al. [46] showed that the diffusion of chloride ions in a mortar can be the result of the diffusion in cement paste plus the geometrical effect of aggregates. Although these results were obtained on Portland cement-based materials, they suggest that the tests on cement paste can be transposed to mortars, the transition from mortar (or cement paste) to concrete being worth exploring, particularly in the presence of SCM.

So, there is a broad range of different models which require different inputs and have different definitions of diffusion coefficients. The available methods for each D at different scales are presented in the following sections.

# 2.2.5 Chloride binding models

Chloride binding in cementitious materials is typically described using linear, Langmuir, or Freundlich isotherm models [23, 27, 47, 48] which relate (using direct analytical expressions) the concentration of bound (or total) chlorides to the free chloride concentration in pore solution. More fundamental approach is based on thermodynamic (numerical) modelling [24–26] that predicts pore solution properties, solid–solution interactions (e.g., hydration, pozzolanic reactions, chloride binding), and supports advanced reactive transport modeling with minimal experimental inputs [48]. Kinetic expressions for chloride binding have also been proposed [47]:

$$\frac{\mathrm{d}C_b}{\mathrm{d}t} = k_b \left( C_{beq} - C_b \right) \tag{3}$$

where  $C_{\rm b}$  is the current bound chloride concentration,  $C_{\rm beq}$  is the equilibrium bound chloride concentration,  $k_{\rm b}$  is the chloride binding rate constant (3.13×10<sup>-7</sup> m³/mol/s). When  $C_{\rm b} > C_{\rm beq}$ , chlorides desorb; when  $C_{\rm b} < C_{\rm beq}$ , free chlorides are bound.

Chloride binding is affected by the w/c, tricalcium aluminate and SCM type, pore solution chemistry, and replacement amounts [48, 49]. Generally, SCM changes the chloride binding; less physical binding [23, 50], with available alumina being considered the factor that leads to the most chemical binding, due



to the formation of greater amounts of Friedel's salt [51]. Kinetic factors and slow evolution of the microstructure must be accounted for in SCM mixtures, and many such corrections (e.g. aging factors) are available in literature [22] [52–55].

#### 3 Diffusion tests

# 3.1 Self-diffusion experiments

Molecular or ionic self-diffusion (Table 3) represents the autonomous movement of molecules or ions within a substance, propelled solely by the inherent randomness of thermal motion. Intraphase diffusion is driven purely thermally, or by specific gradients. In contrast, inter-diffusion arises from concentration (chemical potential) gradients, leading to the blending of different molecules or ions upon contact. Both self-diffusion and intra-diffusion differ from ionic mobility, which results from external forces like an electrical potential gradient. The Nernst-Einstein relationship establishes a connection between the coefficients of self-diffusion and mobility. Diffusion is intricately affected by concentration. Molecular size and interactions, including attractive or repulsive forces, also influence diffusion. Additionally, increased ionic strength impedes diffusion due to enhanced electrostatic interactions among ions.

# 3.1.1 Field gradient nuclear magnetic resonance (FG-NMR) method

FG-NMR provides  $D_0$  (or  $D_{\rm intra}$ ) coefficients of molecules and ions, like chloride, by tracking their random motion using pulsed magnetic field gradients (PFG), without the need for a tracer [4, 56]. These gradients encode spatial information into the nuclear spins. As molecules diffuse, they experience phase shifts in the NMR signal. By analysing signal attenuation with respect to the applied field gradient and diffusion time, the diffusion coefficient is calculated. However, the FG-NMR method is typically limited to the early hydration stages of white cements that are very low in paramagnetic impurities (like Mn- and Fe-oxides).

Nestle [4] reviewed NMR studies on self-diffusion in hydrating cement pastes, noting that diffusion studies are less common due to their complex NMR setup requirements, unlike the more prevalent relaxometry studies with simpler setups. Hansen [57] studied long-range diffusivity by PFG NMR (<1s) and validated it versus  $D_2O-H_2O$  (>h) exchange. The study was done on white cement, and proposed an analytical solution with two D's (within the cement sample and in external solution) to fit the NMR results. Analytical

**Table 3** Definitions and comparison between self-diffusion  $(D_0)$ , intra-diffusion  $(D_{intra})$ , and inter-diffusion  $(D_{inter})$ , in both homogeneous water (solution) and porous media ('inert' concrete)

| Aspect                     | $D_0$ (Self-Diffusion)   | $D_{\text{intra}}$ (Intra-Diffusion)   | $D_{\text{inter}}$ (Inter-Diffusion)   |
|----------------------------|--|--|--|
| Definition                 | Random motion of identical molecules/ions without a gradient                             | Diffusion of a species within<br>a specific phase (e.g., water,<br>solution, pore solution)      | Net movement of multiple spe-<br>cies driven by a concentration<br>gradient          |
| System                     | Single component in homogenous systems   | Multi-component systems, focusing on diffusion in one phase                                      | Multi-component systems, with interaction between species                            |
| Driving Force              | Random thermal motion<br>(Brownian motion)   | Random motion (or specific gradients)  | Concentration or chemical potential gradient   |
| In Homogeneous Water       | Free diffusion of molecules;<br>highest diffusion coefficients<br>due to no restrictions | Like self-diffusion; minor differences due to interactions                                       | Interaction between species (e.g., Na <sup>+</sup> and Cl <sup>-</sup> in saltwater) |
| In Concrete (Porous Media) | -  | Constrained within a specific phase, with additional effects from pore geometry and interactions | Complex mixing behavior across phases; further reduced by confinement                |
| Measurement Methods        | FG-NMR, tracer-free techniques   | FG-NMR, isotopic or chemical tracers (for specific species)                                      | Tracer techniques, diffusion cou-<br>ples, or profile analysis                       |



solutions in [5, 56] considered more common (single D approach) and different diffusion geometries and changing (heavy-water-to-water volume ratio) boundary conditions.

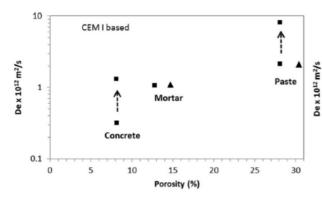
Benefits of NMR methods are in elimination of binding interactions artefacts with cementitious solids, as the binding of proton and deuterium water can be neglected, i.e. it enables the calibration of geometrical pore structure,  $f_p$  in Eq. 2 and 1. NMR measurements, which may take from up to 2 weeks, are still > ~3 times shorter than single (salt) concentration through-diffusion steady-state measurements. However, they are expensive, and FG-NMR measures self-diffusion, which is an average distance travelled by diffusing molecules due to thermal motion in a chemically uniform environment. Inter-diffusion, a more relevant process driven along a concentration gradient involves a tracer (e.g. <sup>2</sup>H or <sup>18</sup>O) diffusion approach, where NMR may be used to measure the concentration changes.

Simple NMR techniques are employed to measure the concentration change during an intra-diffusion process [5]. An inside-out diffusion setup involves following the exchange between water inside the sample with an external deuterium water solution. The intra-diffusion coefficient is obtained by fitting the diffusion model's analytical solution to the measured transient change of the sample's total water concentration. This method is valuable for water-saturated porous media with short relaxation times (<1 ms), making it suitable when conventional NMR pulsed field gradient techniques are impractical. Implementation is straightforward,

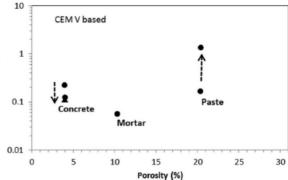
requiring only the recording of a free induction decay (FID) or Hahn echo [4] ,56, 58.

In Fleury's 2020 study [58], diffusion coefficients were examined in CEM I and V paste, mortar, and concrete (with and without metalic fibers) using fast <sup>1</sup>H-NMR. Deuterium-water exchange facilitated inside-out diffusion in cylindrical samples over 17-125 days. Water saturated samples were placed in NMR tubes with deuterium-water. Sample sizes were 10 mm in diameter and 15 mm in height for paste and mortar, and 20 mm in diameter for concrete. NMR magnetization signals were calibrated against known water volumes to measure proton concentration over time. Porosity was determined from additional NMR measurements, following Carr-Purcell-Meiboom-Gill (CPMG) spin-echo pulse sequence approach. The study also compared results with tracer diffusion, Magnetic Resonance Imaging (MRI), and other destructive profiling methods. The pore diffusion coefficient  $(D_n = D_{eff})$ /  $P_{\rm connected}$ ) was calibrated using simple analytical Fick's 2nd solution for inside-out diffusion (analogous to moisture diffusion). Interestingly, porosity was not required to model transient diffusion, as NMR measures total concentration across the sample.

Fleury's study [58] found that CEM V-based materials had significantly lower  $D_{\rm eff}$  compared to CEM I, mainly due to lower  $D_{\rm p}$  and, to a lesser extent, lower accessible porosity (Fig. 2). The addition of fibers increased  $D_{\rm p}$  by creating more diffusive paths, but this effect was counteracted in CEM V concrete by its much lower accessible porosity. The results



**Fig. 2** <sup>1</sup>H-NMR inside-out diffusion method Fleury et al. [58] examined CEM I (left) and CEM V (right) in paste, mortar, and concrete samples. The dashed arrow indicates the inclu-



sion of metallic fibers, while triangles indicate HTO throughdiffusion validation result



324 Page 10 of 35 Materials and Structures (2025) 58:324

showed good agreement with HTO through-diffusion measurements.

Valori et al. 2013 [59] reviewed  $^{1}$ H-NMR experiments on cement-based materials, emphasizing the use of D as a function of diffusion time ( $\tau_{D}$ ) to unveil pore size effects. They cautioned against using the PFG method for gel pores due to size limitations and discussed challenges in cement pastes where adaptations like shorter gradient pulses and specialized sequences are needed to counter magnetic field heterogeneity and mitigate signal loss caused by short  $T_{1}$  and  $T_{2}$  relaxation times of pore water, reflecting rapid dephasing from strong pore surface interactions. Two-dimensional PFG variants could explore relaxation–diffusion correlations and pore anisotropy.

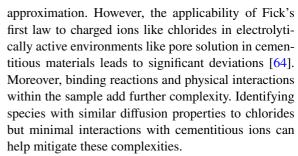
#### 3.1.2 NMR tracer method

The NMR tracer method is a technique used to measure  $D_{\rm intra}$  and  $D_{\rm inter}$  (Table 3) in multi-component systems using NMR-active tracer nuclei (isotopically labelled) [60, 61]. This method involves introducing a small concentration of NMR-active tracer molecule/ions into a system and tracking their diffusion behavior. The tracer's movement is typically monitored using NMR or FG-NMR, to obtain its diffusion coefficient. Although specialized NMR setups are essential for coupling inter-diffusion with drying/capillary sorption, NMR remains effective for simultaneous measurement of moisture and multiple ion contents in (partly moisture saturated) porous materials [62].

# 3.2 Steady-state natural diffusion experiments

Steady state diffusion experiments use two compartments separated by a specimen. The first compartment contains the diffusing species of interest, which penetrates through the specimen into the downstream compartment. The steady-state diffusion coefficient  $(D_{\rm ssd})$  is determined by either elemental profiling along the permeated sample cross-section, measuring the diffusion potential due to charge transfer [63], or by analysing the cumulative species quantity in the downstream compartment. Once a steady increase of cumulative species quantity is reached in the downstream compartment,  $D_{\rm ssd}$  is derived from the linear slope using Ficks's first law [39].

For diffusion of non-charged molecules like tritiated water (detailed below) provides good theoretical



For decades, tritiated water (HTO) has been an elaborate but precise tool for measuring diffusion coefficients of relatively inert, diluted molwithin cement-based materials. involves replacing some hydrogen atoms with tritium atoms (3H) which can be detected due to their radioactivity. Originally used in nuclear power plants, the long-term interaction of Tritium with cement paste (w/c = 0.4...0.55) and mortar (sand/ cement = 1.25...2.25) was investigated [65] using 2.3 cm thin disks (D=4.5 cm). Today, HTO is used for measuring diffusivity in porous materials due to the lack of significant interaction of HTO with the cementitious matrix [61]. Due to long equilibration time (months), the method is limited to (cm) thin samples, thus more appropriate for cement pastes and mortars than concrete samples. The higher variability in mortar results is attributed to specimen heterogeneity due to sand inclusions (90% d < 2.5 mm) [65].

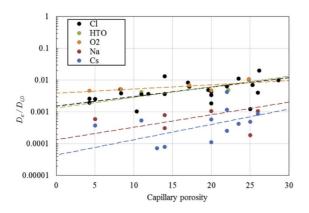
Pure diffusion via HTO is handy and cost-effective but showed that scaling diffusivity tests to cement paste reduces variables, labor and testing time. In [66], comparative results for [39] effective diffusion coefficients of HTO in relation to prominent ions like Cl- or  $O_2$  are analysed from 14 publications. The extensive literature review reveals nearly congruent values for the effective diffusion coefficients of HTO and chloride ions (Fig. 3).

 $D_{\rm eff}$  of chloride ions is similar as that of tritiated water and higher than the sodium ions. This difference can be attributed to the electrical double layer near the charged C-S-H surfaces [66]. The general congruence enables further studies on diffusivity of ions like chlorides but with the benefits of being able to dispense on complex reactivity during diffusion.

Larbi et al. [61] studied the influence of aggregates on the  $D_{\rm eff}$  in mortars using HTO diffusion, confirming the correlation between increased w/c ratio (and aggregate content) and enhanced diffusion. Similar trends were observed in hardened cement paste and



Materials and Structures (2025) 58:324 Page 11 of 35 324



**Fig. 3** Comparison of effective diffusion (relative to self-diffusion,  $D_{\text{eff}}/D_{\text{i,0}}$ ) coefficients for different ions i [66]

mortars [65]. Long-term tracer HTO and Cl36 diffusion [60] showed reduced  $Cl^{36}$   $D_{eff}$  within the first 1.5 years followed by a slower decrease up to 6 years. After 4 years, chloride flux measurements become unfeasible [60] in some cells (1 cm thick: 5 mm cement paste + 5 mm clay; D = 3.5 cm). Sammaljärvi et al. [67] performed similar studies comparing <sup>36</sup>Cl and HTO in two different concretes ( $d_{\text{max}} = 1 \text{ cm}$ ) via autoradiographic imaging of the permeated specimen (h=1 cm; D=3.5 cm) cross-section. They observed similar  $D_{ssd}$  of  $^{36}$ Cl and HTO in both CEM I and CEM V concretes (binding effects should be modelled separately). HTO showed no retardation whereby the distribution coefficient of 36Cl was obvious and pronounced differently between the two kinds of binders in concrete. This study shows the advantage of HTO in cementitious materials to relate the diffusion of an interactive ionic species with the non-reacting species. However, such simplifications are not true in clays. In compacted bentonite [44] HTO accessible porosity was equal to the total porosity, while the chlorides (compared to HTO) show an order of magnitude lower values and a stronger decrease in both  $D_{\rm eff}$  and accessible porosity with increasing dry density. This difference is attributed to the ionic interaction with surface charges bentonite clay.

To sum up, HTO steady-state diffusion tests have negligible reactivity with cement matrices, are suitable for small-scale (mostly cement paste) specimens, correlate well with chlorides, and offer insights into connected porosity. However, HTO behaves differently than charged ions like chlorides in strongly charged solids (like clays), through diffusion setups require long test durations, leading to hydration artifacts, high costs and efforts, and sensitivity to concentration variations [39].

# 3.3 Non-steady-state bulk diffusion experiments

Bulk chloride diffusion testing involves forcing onedimensional chloride ingress into a cement-based specimen sealed on all but one surface and immersed in a chloride-rich solution. After exposure, chloride profiles are measured to calibrate  $D_{\mathrm{nssd}}$   $(D_{\mathrm{app}})$ . Initially, standardized methods such as ASTM C1556 and NT BUILD 443 employed high chloride concentrations (approximately 165 g/L NaCl or ~2.8 M) to accelerate chloride ingress and facilitate rapid comparative evaluation of concrete resistance [7, 8]. In contrast, more recent approaches for calibrating service-life prediction models adopt lower and more environmentally realistic chloride concentrations (typically around 0.3-1.0 M NaCl), closely matching marine splash, tidal zones, and road-deicing conditions [21]. Specifically, RILEM Technical Committee 270-CIM (2022) highlights that chloride ingress models calibrated with these realistic chloride concentrations yield significantly improved predictions of concrete durability in actual service environments compared to those obtained from highly accelerated laboratory tests [21]. This section outlines methods for testing bulk chloride diffusion at both the concrete and cement paste scales, and approaches for measuring chloride profiles in solid phases after exposure. Note that the obtained  $D_{nssd}$  is an average value of the apparent diffusion coefficient over time exposure (not an instantaneous one).

#### 3.3.1 Concrete-scale standardized methods

The standard NT build 443 or ASTM C1556 (Table 1) was developed to measure the diffusion coefficient of concrete using a non-steady-state diffusion test. The test is destructive and time consuming (>35 or>90 days) because it requires the concrete to be grinded or sawn to obtain a chloride profile from titration at different depths. Concrete specimens must mature for 28 days at 20°C, be saturated with limewater, and exposed on one side to 2.8 mol/l NaCl solution (lower concentrations are now commonly employed). After 35+days, concrete is ground in 0.5–1 mm increments to establish a chloride profile from measured



acid-soluble chloride content, excluding the first profile point from regression analysis (because of leaching and surface effects leading to peaking/plateauing behaviours near the surface [23]. More recent ISO 1920–11 and EN 12390–11 use 3 months exposure compared to 1 (to 3) month in NT Build 443, and the variation in exposure time results in differences in the obtained  $D_{nssd}$  [68]. Despite the simplifications [21, 42], Fick's second law is widely used to calibrate  $D_{nssd}$  (and  $C_s$ ), As empirical parameters obtained by fitting the chloride profile with Crank's solution:

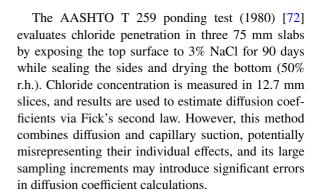
$$C(x,t) = (C_S - C_{background}) erfc \left(x/(4D_{nssd}t)^{-1/2}\right)$$
(4)

where  $C_{\rm S}$  is the surface chloride concentration,  $C_{\rm background}$  the background concentration (initial chloride amount in the samples), x the depth, t the exposure time, and erfc the complementary error function.

A promising new approach for analyzing chloride profiles measured over time on concrete specimens employs the square root law of ffusion [42, 69]. In standard codes (fib), the aging exponent, a lumped parameter (Eq. 1), accounting for hydration-induced microstructure changes and binding non-linearity (RILEM TC-CIM, [22], is derived by regressing  $D_{nssd}(t)$  over exposure periods (e.g., 28, 90, 365, 730 days). A key advantage of this protocol is that it avoids sawing the specimen before exposure to the salt solution, preventing early-age damage. This makes it suitable for studying chloride diffusion during cement hydration. For instance, it has been used to decouple hydration and chloride transport in slagbased mortars [70].

# 3.3.2 Diffusion–advection tests

There are other methods based on transient natural diffusion that also include advection effects. For example, the penetration of chloride ions can be accelerated by external water pressure or capillary suction. Chloride transport can be accelerated by applying pressure to a chloride solution-exposed face, driving advection and diffusion [71]. This method, like water permeability tests, requires a high pressure to be applied and well-sealed sample sides to prevent leakage. While theoretically sound, it is rarely used due to the need for specialized equipment and the risk of microstructure damage from high pressure.



# 3.3.3 Miniature methods adapted for cement paste specimens

No standardized method exists for chloride bulk diffusion test on cementitious pastes, but concrete-scale tests have been scaled-down and adapted for paste specimens. Key differences between scales include casting, curing and chloride profile measurement. Cement pastes are more sensitive to inhomogeneity, bleeding, drying shrinkage and leaching (a focus of the TC 298-EBD working group 1). Homogeneity can be improved with pre-mixing of blended powders [3, 57] and two-stage mixing with water added at each stage [57, 73]. Vacuum mixing suppresses air bubbles [3, 73] and before mixing, water can be deaired [73], deionised [3] or demineralised [73]. Bleeding is a major risk at high w/c ratios, which can result in non-uniform samples, especially in terms of water distribution and microstructure. Bleeding can be mitigated by rotating samples during molding [57, 73, 74] or curing [73] and by introducing a delay in mixing (e.g., 1 h) to allow flocculation of early hydration products (reduce fluidity) [75]. Possible inhomogeneity near the casted surface can be addressed by cutting [3] or grinding [73].

The curing of cement pastes requires special care to minimize leaching. For this purpose, limewater solutions have been used to prevent the leaching of Ca<sup>2+</sup> ions, but such solutions do not prevent the leaching of alkalis [76]. Another possibility is to carry out sealed curing, but there is a risk of desaturation of porosity which will impact hydration and transport. Three approaches are used to avoid a decrease in the sample internal humidity: performing sealed curing with a minimal volume of water [57], using water mixed with a powdered sacrificial sample to limit leaching of the specimen [3] or re-saturating the



Materials and Structures (2025) 58:324 Page 13 of 35 324

sample after curing and pre-conditioning [73]. The minimal water volume approach is also supported by a recent study on the effect of curing of cement pastes on microstructure and electrical resistivity, which showed that it induces the least effect on paste properties while avoiding problems related to leaching and drying (e.g., cracking, incomplete re-saturation, etc.) [77]. The curing duration is also a variable between studies (e.g., from 28 days to 6 months). When possible, a longer curing time allows for the study of a more stabilized microstructure, resulting in a more constant diffusion coefficient over time.

Once curing is complete, a chloride-impermeable coating is applied to all surfaces of the samples except one to achieve one-dimensional diffusion. This coating is typically epoxy-based, although Jensen et al. [73] suggested the use of polyurethane. This coating can be applied on freshly cut surfaces [59, 73], or on dried surfaces [57]. If the coating process results in drying of the specimen, resaturation prior to exposure will prevent chloride penetration by advection [57, 73] as also performed at the concrete scale (e.g., ASTM C1556).

The NaCl concentration of the exposure solution varies from study to study. In most cases, the solution is prepared to simulate seawater at 3% (0.51 mol/l) NaCl [3, 59, 78] or 3.9% (0.67 mol/l) [57], but other authors also used a lower concentration of 0.115 mol/l of chloride to simulate groundwater [74]. Leaching during exposure also has an important effect on the measurement. The choice can be made to suppress the effect of leaching [73] or to include its impact by renewing regularly the exposure solution [3, 59]. The ratio of the exposed surface of the sample to the volume of the exposure solution is another important parameter for leaching; it is usually set at 50 cm<sup>2</sup>/L, as specified in the ASTMC1556 standard [3, 66, 79]. Although this ratio has an important effect on the properties of the sample and the results, it is not always reported. The temperature is also an important parameter to report but it is not always indicated.

After exposure, chloride profiles are measured using either chemical or spectroscopic techniques (see the following section for more details). For the concrete scale, a calibration of Fick's second law of diffusion can be used to estimate  $D_{app}$  or  $D_{\rm nssd}$ . However, analyses on cement pastes have the advantage of much higher spatial resolution (e.g., a chloride content measurement per mm), which allows better

identification and understanding of the surface effects leading to peak or plateau surface behavior [23]. Additionally, controlled exposure conditions and precise measurements allow studies to be performed after much shorter exposure times, e.g. 1 day [79, 80].

Similarly, while the square root law for chloride diffusion was initially developed for analysing long-term concrete exposure [69, 81, 82], its application to cement paste samples has enhanced understanding of chloride ingress behavior [3, 42, 80, 84–86]. For example, Wilson et al. studied transport and binding properties of various blended cement systems exposed to seawater-like chloride concentrations, showing that penetration is primarily diffusion-driven, with little effect of the very different binding behavior, i.e., chloride penetration obeys the square root law with an offset to account for microstructural changes that occurred prior to the first measurement [3, 50].

Moreover, the penetration depth at a reference chloride content divided by the square root of time  $(X_{Cr}/\sqrt{t})$  turns out to be a good indicator of the resistance against bulk chloride diffusion and correlates very well with  $D_{app}$  [3, 83]. Further research is needed to strengthen the use of square root law to model chloride diffusion, but this can be very promising as measurements could be simplified by focusing only on penetration depth rather than full profiles.

Fitting chloride profiles with Crank's solution to Fick's law provided apparent diffusion coefficients  $(D_{\rm nssd})$  and surface chloride concentrations  $(C_{\rm s})$ , summarized in Table 4. Across all systems, the differences between SEM–EDS and grinding titration (GT) methods remain within ASTM C1556 limits ( $\leq$ 39.8% for  $D_{\rm nssd}$ ,  $\leq$ 37.2% for  $C_{\rm s}$ ), indicating acceptable intermethod variability. The observed differences  $(d_{\rm 2s})$ , while sometimes high (e.g., 32% for  $D_{\rm nssd}$ ), are typical due to the sensitivity of the fitting equation to slight profile variations. SEM–EDS consistently yields slightly higher values than grinding titration and is better suited for short-term exposure measurements due to its superior spatial resolution.

# 3.3.4 Methodology for chloride profile measurements in the solid phases

Measuring chloride profiles after bulk diffusion tests can be done by different methods based on five principles: chemistry-based techniques, X-ray spectroscopy,



324 Page 14 of 35 Materials and Structures (2025) 58:324

**Table 4** Effect of SCMs (GGBFS=Slag, Fly Ash=FA, Calcined Clay=CC) on Fick's 2nd solution fit parameters D<sub>nssd</sub> and C<sub>s</sub> for chloride concentration profiles measured by SEM-

EDS and grinding titration (GT) after 100 days of exposure in 0.5 M NaCl. The difference between the SEM-EDS and GT methods is indicated as d2s

| Paste mix (w/b=0.4)    | $D_{\rm nssd~EDS}$ , $10^{-12}$ m <sup>2</sup> /s | $D_{\text{nssd GT}}$ , $10^{-12}$ m <sup>2</sup> /s | d <sub>2s,</sub> % | $C_{\text{sEDS}}$ , wt% | $C_{ m s~GT}$ , wt% | d <sub>2s,</sub> % |
|------------------------|---|---|--------------------|-------------------------|---------------------|--------------------|
| 100%OPC                | 11.36   | 8.23  | 32.0               | 1.78                    | 1.80                | 1.1                |
| 100%PLC<br>(30% CaCO3) | 11.81   | 14.15   | 18.0               | 0.53                    | 0.50                | 5.8                |
| 70%PLC+30%GGBFS        | 1.12  | 0.90  | 21.8%              | 1.65                    | 1.84                | 10.9               |
| 70%PLC+30%FA           | 3.76  | 3.70  | 1.6%               | 3.21                    | 2.25                | 35.2               |
| 70%PLC+30%CC           | 0.42  |   |                    | 1.88                    |                     |                    |

X-ray absorption, plasma spectroscopy and mass spectroscopy. These methods are generally easier to apply to cement paste samples, but they can also be used for concrete by either identifying its individual components or assuming that only the paste fraction contributes to chlorides binding. The methods are briefly described here in the context of obtaining chloride profiles, but further review of each method itself can be found in a previous review on measuring chloride binding in cementitious materials prepared by RILEM TC 298-EBD [86].

In concrete, chlorine is mainly present in the form of chloride, either dissolved in the pore solution or bound in salts and hydrated phases. However, due to their physical measurement principles, many analytical techniques detect elemental chlorine. For simplicity, the term chloride is used throughout this section, even when referring to measurements based on chlorine.

3.3.4.1 Chemistry-based techniques Currently, the most used technique is the chemical titration of ground powder (ASTM C1152, ASTM C1218, AASHTO T 260–21, NT Build 208, [11–13]. Sample is sliced into thin layers, ground into powder, and chlorides extracted using nitric acid or water. Chloride concentration is then measured by potentiometric or Volhard titration method [86].

Acid dissolution measures total chloride, while water extracts free and some bound chlorides [87]. Bonnet et al. [88] quantified the source of uncertainties for total chloride profiles measurements by considering human, instrument and protocol errors. The grinding/titration method, though common, has limitations: labor-intensive, error prone [89], low spatial resolution (1–2 mm) and variable accuracy depending

on pH [90]. It works best for cement paste but requires large samples for mortar and concrete due to chloride dilution and lower spatial resolution.

Colorimetric measurement using AgNO<sub>3</sub> spray [91] on a cross-sectioned specimen forms a silvergray AgCl precipitate in chloride-rich regions [92], allowing visualisation of the chloride penetration front and determination of the characteristic concentration  $C_d$  (the chloride concentration corresponding to the colour change at penetration depth  $x_d$ ). It is fast and simple but depends on pore solution alkalinity [93],  $C_d$  varying between 0.07M and 0.7M (free chloride) in different mixes [93], limiting its reliability across blended binders [94]. If Cd and surface chloride are known, diffusion coefficient can be estimated [3, 7]. These  $C_d$  values should be discussed in the future to better assess the value of  $D_{app}$  in the case of concrete with SCM. Moreover, distinguishing color changes is difficult, especially in dark or low-penetration samples. Modified methods using K<sub>2</sub>CrO<sub>4</sub> or fluorescein [95] improve visibility. Downscaling the study to cement paste mitigates heterogeneity issues due to aggregates [96].

Due to the limitations of the titration of chloride and the colorimetric techniques, alternative techniques were developed. Some of these methods rely on the excitation or ionization of atoms using an external energy source, causing them to emit characteristic X-rays as they return to a lower energy state. Different energy sources can be used: X-ray beam for µXRF [97], electron beam for Energy Dispersive X-ray Spectroscopy (EDS) or Electron Probe MicroAnalyser (EPMA) [98–101] and particle beam (PIXE) [102]. These techniques allow characterising chemically the sample and creating maps of the chemical composition, which can be divided into



Materials and Structures (2025) 58:324 Page 15 of 35 324

slices and the mean chloride content of every slice is used to obtain the profile. A more detailed review of the methods can be found in the previous TC 298-EBD publication [86].

Depending on the source of energy, the resolution, and the limit of detection of chloride content varies. These values are given in the Table 5. In most cases, the spatial resolution is improved compared to the titration so that the profile obtained is more precise thanks to the smaller sample volume under investigation.

Another advantage of these techniques compared to titration is the ability to map the chloride concentration, so that only the cement paste is studied in mortar samples after excluding aggregate volumes from the measurement data using criteria on calcium and silica contents [89, 100].

In the case of EDS and EPMA, the sample needs to be cut, impregnated in epoxy, polished and coated with carbon to evacuate charge. These manipulations are risky because it can be a source of contamination of the sample with chloride, displacement of chloride or carbonation products. To prevent these risks, ethanol is used as a lubricant for both cutting and polishing [99] but it can also be a dry cutting [92]. Bernard and Wilson [103] proposed a simplified protocol using variable pressure SEM-EDS to quantify chloride ingress without complex calibration or extensive sample treatment. The procedure measures chloride profiles at various depths across  $\sim 200 \times 900 \,\mu\text{m}^2$ areas, large enough to average local variations and minimize surface preparation effects. Depth intervals ranged from 300 µm to 2.5 mm, enabling fine resolution near the surface and efficient analysis for deeper ingress. The measured concentration was assigned to the central depth of each area, with image analysis used to determine the exposed surface. Varying the chamber pressure between 0.1 and 10 Pa showed no significant effect on quantification. A strong correlation ( $R^2$ =0.993) between SEM-EDS at 0.1 Pa and grinding titration confirms the method's quantitative validity. Differences between methods were under 10%, indicating good agreement. Chloride profiles measured on OPC and LC3 blends specimens after 36 days matched ASTM C1556 criteria for repeatability, validating the proposed SEM-EDS protocol.

For the  $\mu XRF$  measurement, no coating is necessary, and the polishing is less sensitive thanks to the greater depth of penetration of X-ray in the sample. Finally, the PIXE does not require any sample preparation. Calibration of spectroscopy methods can be useful, but not necessary [3].

Laser-Induced Breakdown Spectroscopy (LIBS) is another technique for estimating chloride content in cementitious materials [104]. In this method, a high-energy laser ablates the sample surface, generating a plasma that emits light analyzed spectroscopically to determine elemental composition. LIBS is an established method for chloride analysis in concrete, as outlined DGZfP bulletin B14 [105]. Laboratory studies have demonstrated its sensitivity, with detection limits for chloride in cement paste samples averaging approximately 0.09 wt.% [106]. For accurate quantification, however, matrix-matched calibration samples are required [107].

Sample preparation for LIBS is minimal, requiring only the splitting of concrete cores, and measurements can be performed directly on rough surfaces [105]. The technique enables spatially resolved measurements, preserving critical localized information

**Table 5** Comparison of the chemical titration of ground powder with other Cl profile measurement methods

| Technique          | Detection limit (wt.% cement) | Spatial resolution          | Notes  |
|--------------------|-------------------------------|-----------------------------|--|
| Chemical titration | < 0.01 [117]                  | 1–2 mm [7]                  | Standard, widely used  |
| Colorimetry        | /                             | /                           | Less detailed info   |
| EPMA               | 0.01 [99]                     | 0.5 μm [100]                | High spatial resolution  |
| EDS                | 0.1–0.2 [118]                 | 37 μm [118]                 | Suitable for fine profiling [103]; validated SEM–EDS protocol* |
| μXRF               | 0.1 [89]                      | 50 μm [ <mark>89</mark> ]   | Moderate resolution  |
| LIBS               | 0.09 [106]                    | 100 μm [ <mark>119</mark> ] | Laser-based technique  |
| LA-ICP-MS          | 0.12 [120]                    | 300–400 μm [120]            | Sensitive but coarser resolution                               |
| TXM                | /                             | 8.8 μm [121]                | X-ray microscopy method  |

\*Variable pressure SEM— EDS with minimal sample prep and no complex calibration [103]:



324 Page 16 of 35 Materials and Structures (2025) 58:324

about chloride distribution such as in cracks, which would otherwise be lost through sample homogenization. The spatial and local resolution of LIBS depends on the specific system used and can achieve scales below 100 µm or even finer. Additionally, LIBS facilitates simultaneous multi-element analysis, enabling the selective analysis of the binder matrix in concrete samples by identification of individual concrete components [108–110], thereby providing a direct correlation between chloride content and the binder phase. Its reliability for chloride analysis is further supported by good agreement with acid-soluble methods [111]. Moreover, the availability of remote and portable LIBS systems expands its utility for rapid field measurements [112].

A key advantage of LIBS over many other analytical methods is its ability to detect light elements, including hydrogen, providing a broader element analysis capability. Although measurements can be performed under atmospheric conditions, signal quality can be enhanced by introducing an inert gas around the plasma [104], dual pulse configuration [113], electric sparks [114], or microwave coupling [115]. Despite its considerable potential, LIBS remains underutilized within the cement community, with its commercial application largely confined to a few specialized laboratories, primarily in Germany and Switzerland.

LA-ICP also utilizes the plasma state of the sample for chemical characterisation and can therefore be used to measure spatially resolved composition [74]. The sample undergoes laser ablation, transported in argon, and transformed into plasma by induction. The ions created are chemically segregated by a mass spectroscopy according to their mass to charge ratio. Measurements can be performed directly on solid surfaces, but accurate quantification requires careful system calibration. Finally, it has a particularly good resolution and limit of detection (see Table 5).

Transmission X-ray microscopy (TXM) technique uses the difference in absorption of X-ray between a cement paste with a high concentration of ions and another one with a low concentration of ions. Because of the weak ability of chlorides to absorb X-ray, they are replaced for this test by iodides that are of comparable size and more absorbent. The diffusion test is made with a solution of KI and then X-rays are transmitted through the sample. The profile of iodide is then obtained with the profile of absorption of the

sample. It is possible to make a calibration of the test to obtain quantitative values. The main drawback of this technique is that it does not use chloride but iodide that has a different size, ionic mobility and binding ability so the diffusion coefficient varies slightly compared to that of chloride [116]. The possible differences in behavior, especially for liquid/solid interaction, makes this method not adapted to diffusion with binding. Mortar or concrete samples make the measurement more complex because iodide shows nonuniform penetration when the solution reaches large features like aggregates.

3.3.4.2 Comparison of the methods with potentiometric titration In addition to the well-established approach of sample grinding followed by potentiometric titration, several alternative analytical techniques are available for the quantification of chloride in concrete. These methods offer various advantages, notably in terms of sample preparation: most can be applied directly to freshly split or polished core surfaces, eliminating the need for extensive sample preparation.

Surface scanning techniques, in particular, provide improved spatial resolution, enabling a more accurate assessment of chloride ingress, especially in the presence of cracks. Moreover, they facilitate the selective analysis of distinct concrete components, allowing for a direct correlation between chloride content and the binder phase.

All methods considered offer detection limits sufficient for identifying critical chloride concentrations (0.2–0.5 wt.% per binder), as summarized in Table 5. However, accurate quantification often requires the use of matrix-matched calibration standards. In addition, some of these techniques support in-situ measurements with portable or remote-operable devices.

- o Depth intervals: 300 µm to 2.5 mm
- Validated repeatability per ASTM C1556 for OPC and SCM blended cement pastes

# 4 Migration tests

Several types of rapid chloride penetration tests have been developed over the last decades. Common to all is that a concrete sample is placed adjacent to a chloride solution on the one side and a chloride-free



Materials and Structures (2025) 58:324 Page 17 of 35 324

solution on the other. A potential difference is then applied to drive chloride ions through the concrete. A chronological review of reported test methods is first presented in this section, followed by specific descriptions of current steady-state and non-steady-state migration methods for concrete and cement pastes.

# 4.1 Historical development of migration methods

The first type of rapid test is represented by the AASHTO T277 test [122], developed by Whiting [123], in which the total charge passed through a sample during a six-hour period under a 60 V potential difference is measured, and the value is used as a chloride permeability index. The second type of test reported is developed by Dhir et al. [124], who related the measured steady state chloride flux under a 10 V potential difference empirically to the chloride diffusion *D* of the concrete. The *D* values from Fick's first law exceeded conventional natural diffusion coefficients, making it an index of chloride diffusion rather than a true *D*.

The third type of test is represented by the developments of Tang and Nilsson [125], further standardized as NT-Build 492, in which the depth of chloride penetration under a 30 V potential after 8 h is measured. In their proposed method the authors established a relationship between the penetration depth of chloride ions and a chloride diffusion coefficient  $D_{\rm nssm}$ , based on the measured  $c_d$ , the chloride concentration at which the colour changes when a colorimetric method is used to measure the penetration depth  $x_d$ .

In NT Build 492 [16] and NF XP P18-462 [17] the diffusion coefficient calculated through this method is defined as  $D_{\rm nssm}$ , reflecting that this value is calculated through a non-steady state migration transport process. Even in the case of the method proposed by Dhir et al. [124], although Fick's first law of diffusion was used for calculation of the index of chloride diffusion, the test set-up was not following a steady state transport regime. Andrade [126] presented a theoretical steady-state migration test method suggestion, which was further experimentally tested and presented in (Andrade et al., 1994) [127] and modified to become the standard NF XP P18-461 (details are provided in Sect. 4.2).

Tang [128] also suggested a steady-state method which was adapted from the method by Whiting

[123]. Andrade [126] proposed a steady-state migration test method, later further experimentally validated [2, 129].

As stated in [128] the greatest difference between the steady-state migration and non-steady-state migration is that in the former the chloride ions must pass through the specimen and the flow rate must be measured until steady-state flow is obtained. Another difference is the destructive aspect of the non-steady state migration coefficient. Indeed, the specimen could be used after the test for other measurements as done by Djerbi el al. [130, 131] to evaluate the influence of damage on chloride ingress. Given the longer time duration of the method, durable anodes should be used such as platinum to maintain a clean anolyte solution, while evaporation in anolyte should be prevented. Given the experimental duration, thinner discs are suggested (30 mm) than in the case of non-steady-state migration (50 mm). A lower applied potential (24 V) compared to non-steady state migration (30 V) is also recommended to keep the electrical field to 600 V/m. A high concentration of chloride in anolyte should be avoided as the produced hydration phases might cause pore blockage and consequently decreased flow. A solution of 3% NaCl in 0.1 M K/Na (OH) solution is suggested. To prevent production of chlorine gas the pH value in anolyte should be kept to more than 12. Therefore, a 0.3 M K/Na (OH) solution and frequent control of the pH in anolyte are important [132]. However, controlling the chloride concentration in the downstream cell is challenging due to limitations in measurement accuracy and potential chemical reactions at the anode. Therefore, further improvements of the method were suggested by Truc et al. [133], in which instead of controlling the concentration in the down-stream cell, a control for upstream cell was suggested. Further, Castellote et al. [134], presented an update in the method in which monitoring of the conductivity of anolyte chamber was suggested instead of concentration measurements. Their work is an attempt to also account for chloride binding by means of calculating the time-lag taken by the chlorides to appear in the downstream or anolyte chamber.

It is, however, important to not underestimate the effect of migration on pore solution characteristics of the cementitious sample through the migration test. As presented by Cherif et al. [135], the ionic concentrations change in the sample during a migration test.



Decomposition of portlandite, calcium release and the decrease of Na<sup>+</sup> and K<sup>+</sup> in the sample pore solution are important aspects not usually considered in literature. Moreover, as presented by Hemstad et al. [136], the capacity of the paste to bind chlorides is a complex function which can be affected by phase assemblage and chloride concentration as a function of depth, pH and leaching. These outcomes infer that the calculation of bound chlorides through a migration test might have inaccuracies assigned to it.

# 4.2 Steady-state migration experiments

NT Build 355 and NF XP P18-461 background is introduced in the previous section. Its main drawbacks are discussed in following section.

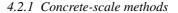
To estimate D from migration measurements in concrete, three key factors are considered: I) Experimental data collection—monitoring chloride flux (J) over time while applying a suitable potential difference  $(\Delta E)$  to drive migration vs minimizing the Joule effect. 2) Solving the Nernst-Planck equation—relating flux to  $D_{ssm}$ , simplified as (see assumptions below):

$$D_{ssm} = JRTL/(zFC_{cl}\Delta E)$$
 (5)

or alternatively to 2, 3) Using the Nernst-Einstein equation—linking flux to total current intensity (*i*) and the transference number  $(t_j)$  of migrating anions, yielding:  $J = (i t_j) V(n F)$ . By calculating the transference number  $(t_j)$  from the experimental data and substituting it into the simplified equation to obtain  $D_{ssm}$ :

$$D_{ssm} = RTit_{cl} / \left( nF^2 \Delta EAC_{cl} LZ \right) \tag{6}$$

where L is the specimen thickness. The framework for calculating  $D_{\rm ssm}$  (NT Build 355 introduced 1997), considering steady-state flow, potential difference, and ionic strength in concrete pore solutions is based on following assumptions. No advection, diffusion is negligible at  $\geq$  10V, chloride concentration is constant upstream, and the electric field decays linearly across the specimen. Ionic mobility is higher in solution than in the sample (L is the transport distance). The mathematical equations from Andrade and Tang [126, 128, 129] were similar with a minor difference being Andrade's [126] inclusion of chloride ion activity, which Tang excluded due to dominance of strong external electrical current.



The ASTM C1202 'Coulomb test' measures charge passed (based on Nernst-Planck and flux related to transference number, see previous section) as an index of chloride resistance but cannot directly determine *D* due to the involvement of all ions [137]. However, using almost the same setup, a steady-state diffusion coefficient can be theoretically calculated, as outlined in NT Build 355 [16] (Table 1). The chloride migration coefficient can be determined by measuring the electrical current over time and analyzing it with the Extended Nernst-Planck model (Samson 2003) [138], while other species' migration coefficients are computed based on their diffusion coefficient ratios relative to chloride at infinite dilution.

However, in the experimental setup chloride ions may convert to chlorine gas in the downstream cell, which is difficult to control and quantify. Additionally, the electrical potential between electrodes, rather than across the specimen, is used in calculations, ignoring the potential drop at the specimen surface. The model also assumes a single species theory, neglecting interactions with other ions. To address the single specie limitation in obtaining  $D_{\rm ssm}$ , Zhang et al. [139] introduced a correction factor  $(\beta_0)$  dependent on (NaCl) concentration and temperature. Zhang et al. [140] developed an electrochemical approach linking observed migration rates to  $D_{eff}$ , while NF XP P18-461 refined Andrade's methods, using potentiometric titration to demonstrate equivalence between  $D_{\it eff~(migration)}$  and  $D_{\it ssm}$ Castellote et al. [134] presented a faster migration test by monitoring downstream (anolyte) conductivity, though it excluded  $D_{\rm ssm}$  determination. Truc et al. [133, 141] proposed strategies to address limitations in steady-state methods, focusing on pore structures and assuming the upstream flux (being constant from the test start and independent of chloride binding). Together, these works advanced rapid chloride diffusion assessment in concrete. Neglecting electroneutrality in NT Build 355 underestimates D; an order of magnitude compared with a Friedmann's steady state migration method based on current measurement and Nernst-Planck (with electroneutrality) analytical model (Friedmann, 2004) [142]. KrabbenhØft et al. [143] used coupled Nernst-Planck-Poisson model for migration tests and found effective diffusivity differences of 50–100% compared to single-species models



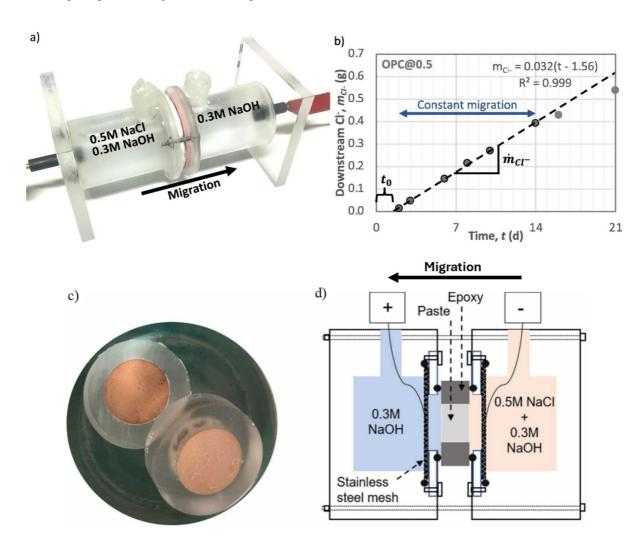
like NT Build 492 and NT Build 355. KrabbenhØft suggested adjusting the NaCl/NaOH ratio in cells to improve results with simplified models.

Finally, the unclear behavior of chloride ion binding under an electric field may introduce systematic errors, highlighting the need for further research across different current densities.

# 4.2.2 Miniature methods for cement pastes

While most studies have been carried out on concrete specimens, researchers have also developed test setups to perform migration tests on paste-scale

specimens, sometimes referred to as "mini-migration tests" (2004) [75, 144, 145] (Fig. 4). The parameters and differences between these methods are presented in Table 6. Note that less experimental details have been reported by Hu et al. [145] which limits the comparison with the other reported methods. They used kaolinitic clays mixed with Singapore's low-grade marine clays (16–20% kaolinite and illite and quartz as major phases). When used in LC<sup>3</sup> systems, these clays reduce 28-day strength by 25% compared to OPC, but promote CO<sub>3</sub>-AFm formation, improve chloride resistance by an order



**Fig. 4** Wilson et al. [75]: a) A mini-migration setup drives chlorides through a 33 mm $\times$ 10 mm cement paste disk. b) For OPC (w/b=0.5), AgNO<sub>3</sub> titration of downstream samples shows steady chloride migration after breakthrough ( $t_0$ ), last-

ing until reservoir concentrations shift (t>14 days). Hu et al. [145]: c) Cement paste samples flat cylinders with epoxy covering the lateral sides; d) mini-migration set-up scheme



324 Page 20 of 35 Materials and Structures (2025) 58:324

Table 6 Comparison of the miniature steady state migration methods for cement pastes

| Method                   | NT Build 355  | Ait Mokhtar et al. [144]         | Wilson et al. [75]               | Hu et al. [145]             |
|--------------------------|---|----------------------------------|----------------------------------|-----------------------------|
| Curing                   | Water at 20±2 °C  | Downstream solution              | Sacrificial solution             | Sealed curing               |
| Curing time              | 91 days   | 28 days                          | 28 days                          | 28 days                     |
| Pre-saturation           | Yes   | No                               | Yes                              | Yes                         |
| Polishing                | No  | No                               | Yes                              | Yes                         |
| Core sampling            | Yes   | Yes                              | No                               | No                          |
| Sample dimensions        | 95-100 mm diameter and 50 mm thick                                | 63 mm diameter & 10 mm thickness | 33 mm diameter & 10 mm thickness |                             |
| Voltage                  | 12 V  | 300  V/m = 3  V                  | 5; 12 V                          | 5; 12 V                     |
| Reservoirs sizes         | Measured before test  | 1 L downstream, 2 L upstream     | 125 mL each reservoir            | 113 mL                      |
| Downstream solution      | 300 mM NaOH   | 25 mM NaOH+83 mM<br>KOH          | 300 mM NaOH                      | 330 mM NaOH                 |
| Upstream solution        | 860 mM NaCl   | Downstream + 500 mM<br>NaCl      | Downstream + 500 mM<br>NaCl      | Downstream + 500 mM<br>NaCl |
| Solution renewal         | To maintain the concentration in the upstream reservoir           | 10–15 h                          | None                             |                             |
| Test duration            | >7 days   | 14 days                          | 2-3 weeks                        |                             |
| Measured quantity        | Voltage drop and down-<br>stream Cl content                       | Current & downstream Cl content  | Downstream Cl content            | Downstream Cl content       |
| Frequency of measurement | At least each days  | N/A                              | 2–4 days                         | 1–3 h                       |
| SCM effects              | $\downarrow D_{ssm}$ mostly due to refine tivity (not normalised) | ement of pore structure by S     | SCM (Fig. 5); linear correla     | tion with bulk conduc-      |

of magnitude (Fig. 5a), and refine pore structure despite increased porosity.

The proposed methods are down-scaled adaptations of the AASHTO T277 test setup [122], developed by Whiting [123]. The current and/or chloride concentration are measured over time and a diffusion coefficient is calculated based on the approach of Andrade [129]. As with the diffusion tests, when scaling down to the cement paste scale, special attention must be paid to possible problems with sample preparation, bleeding, drying, leaching and more.

Wilson et al. [75] observed a strong correlation between  $D_{ssm}$  and bulk conductivity  $\sigma_b$  (Fig. 5b) for a wide range of SCM (CC, LL, GGBFS, FA, glass powder) and w/b ratios, supporting that both tests can characterize the effective chloride ingress, with  $D_{ssm}$  being useful in transport models. They thus conclude that bulk conductivity is a good indicator for chloride ingress resistance and that it should not be normalized by the pore solution conductivity due to the impact of the pore solution on diffusion. In addition, the use of the mini-migration setup allows in-depth analysis

and understanding of migration-related phenomena, as complementary experimental techniques can be used to examine the exact same cement paste samples before, during and after mini-migration, in terms of microstructure (e.g. with mercury intrusion porosimetry), phase composition and chemically bound chlorides (with XRF and XRD), elemental and phase profiles (with SEM–EDS), together with the composition and conductivity of the pore solution. Such coupled analyses allowed to distinguish the mode of action of different SCM on chloride diffusivity (i.e. pore network properties vs. pore solution conductivity) [146].

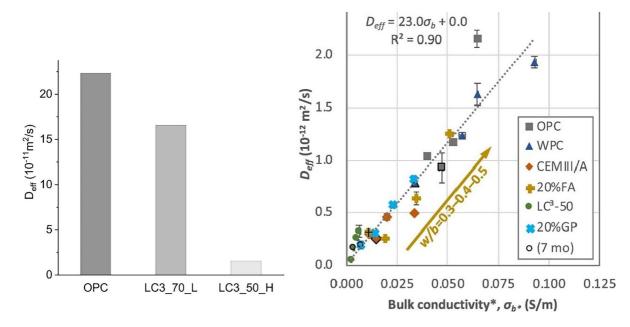
# 4.3 Non-steady-state migration experiments

#### 4.3.1 Concrete-scale standardized methods

The NT BUILD 492 method [16] is a widely used rapid non-steady-state migration test for assessing chloride penetration resistance in concrete, mortar and cement-based repair materials [147]. This method relies on chloride ion migration under an external



Materials and Structures (2025) 58:324 Page 21 of 35 324



**Fig. 5 a** Hu et al. [156] results of Dssm (=Deff; LC3\_70\_L has 71% OPC +18 % low grade 3 mixed calcined clay; and LC3\_50\_H has 53% OPC + 15 % metakaolin rich and 15% low 4 grade mixed calcined clay). **b** Wilson et al. [78] linear

correlation between Dssm (=Deff) and 5 bulk conductivity taken as the average between 28-day and 91-day results to consider 6 microstructure changes occurring during the minimigration test

electrical field, using a potential difference and a colorimetric indicator (0.1 M AgNO3 spray). It provides quantitative data, including chloride penetration depth and diffusion coefficient, with test durations ranging from 24 h for ordinary concrete to 96 h for high performance concrete [148]. In NT BUILD 492, specimens (100 mm diameter,  $50 \pm 2$  mm thick) are sliced from cast cylinders or drilled cores with a minimum length of 100 mm. NT Build 492 (the basis for EN 12390-18 and XP P 18-462) involves specimen presaturation pre-treatment (under vacuum and with a Ca(OH)<sub>2</sub> solution), using a 10% NaCl upstream solution. In contrast, EN 12390-18 does not use vacuum pre-saturation and samples are cured in water without Ca(OH)<sub>2</sub>, but the method uses an upstream alkaline solution, i.e., 5% NaCl in 0.3 M NaOH.

A DC potential of 30 V is applied, adjustable between 10 to 60 V to limit specimen power consumption to less than 2 W. After the test, the specimen is split and chloride ion penetration depth is determined using a colorimetric method [76]. Chloride penetration depth is measured every 10 mm on each split piece, yielding 5 to 7 valid readings per specimen. To ensure safety and minimize contamination, AgNO<sub>3</sub> spraying should be performed in an

enclosed space. The chloride diffusion coefficient  $(D_{nssm})$  is calculated as:

$$D_{nssm} = \frac{RTL}{zF\Delta E} \frac{x_d - \alpha\sqrt{x_d}}{t} \tag{7}$$

where  $x_d$  is the average value of the penetration depths, t is the test duration.  $\alpha$  is a laboratory constant (Eq. 8):

$$\alpha = \frac{2}{\sqrt{a}} erf^{-1} \left( 1 - \frac{2C_d}{C_0} \right) \tag{8}$$

The NT BUILD 492 offers good precision, with a repeatability COV of 5–9% and a reproducibility COV of 12–24% (13% for Portland cement or silica fume concrete, and 24% for slag cement concrete) [149]. It is a simple, rapid method with straightforward calculations, producing results comparable to the longer NT BUILD 433 tests [149]. However, the theoretical model has notable oversimplifications [150]. It assumes independent chloride ion movement, neglecting interactions with other ions in the pore solution [150]. The method also presumes a constant electrical field and disregards chemical



reactions with the hydrated paste [138, 147]. Moreover, the accuracy of colorimetric method is questionable, as the *Cd* value is affected by SCM due to their effect on pH. This point needs more introspection in the future.

Tang et al., noted significant depth measurement errors at shallow depths, though errors at depths over 10 mm can be reduced to a few percent [151]. To minimize specimen heating, the increase in applied voltage should be controlled, and the volume of the salt solution increased [152]. Excessive potential difference can lead to overheating, altering results, and damaging concrete specimens [153].

As shown by Hemstad et al. [136], chloride binding capacity depends on phase assemblage, concentration, depth, pH and leaching. The unclear behavior of chloride ion binding under an electric field may thus introduce systematic errors. Additionally, the kinetics of binding, both during and after the application of the voltage, warrant investigation. Specifically, delays in conducting profile tests could influence ongoing binding reactions, particularly if equilibrium has not yet been reached. Another major drawback of many electrically accelerated methods is their reliance on single-species theory, which neglects the influence of other ions. While coupling the Nernst-Planck and Poisson equations improves theoretical accuracy, it remains too complex for practical engineering use. Friedmann's analytical solution [142], which incorporates the Nernst-Planck model and electroneutrality, shows promise, but further efforts are needed to make the experiment more practical and applicable to nonsteady-state conditions.

Finally, one should keep in mind that in durability prediction models, using input D parameters obtained by migration methods is considered as an empirical approach [52, 154] where  $D_{\rm nssm}$  should always incorporate a calibration factor to account for the correlation with  $D_{\rm app}(t)$  [68, 155].

# 4.3.2 Miniature $D_{nssm}$ methods for the cement paste scale

Huang et al. [156, 157] modified the NT BUILD 492 to evaluate chloride migration in pastes blended with fly ash, slag and limestone. The primary adjustment for paste was reducing specimen size to prevent cracking, using rubber cylinder moulds with a  $50\pm2$  mm diameter (half that of concrete samples) and

110 mm length. Pastes were sealed cured for 1 day, demoulded, and water cured until the target ages (e.g. 28, 90, 180 days). For RCMT, two  $50\pm2$  mm thick samples were cut from each specimen, and a preset voltage of 40 V was applied to determine the initial current. Penetration depth was measured at~5 mm steps after the white silver chloride precipitation was visible on split surfaces, following NT BUILD 492 procedures. Similarly, Ranger and Hasholt (2023) [158] used smaller samples (22 mm diameter) for chloride migration tests, measuring only two depths per specimen to determine the penetration depth (Table 7). The calculation of  $D_{nssm}$  for pastes followed the same equations used for concrete samples. Table 7 compares the key setup differences between standard concrete tests and these customized paste methods. Practical challenges include potential sample breakage or accelerated dissolution at the interface between clamped and unclamped regions after testing, especially if the applied current is too high. This can complicate the accurate measurement of penetration depth.

The miniature methods at paste scale effectively quantify the influence of SCM on the chloride migration in cement-based materials. Huang et al. (2022a) [157] demonstrated that the major cause of changes in migration coefficient of blended pastes is the refinement of pore structure by SCM. This refinement can be assessed by the electrical conductivity properties of cement-based materials [159]. Ranger and Hasholt [158] found that the  $D_{\text{nssm}}$  of paste correlated with the reactivity of the SCM (Fig. 6), and the inverse of formation factor (1/F), where formation factor F is the sample conductivity normalized by the conductivity of the bulk pore solution). Moreover, they found a clear correlation between paste and concrete properties, regardless of binder type, though aggregates affect  $D_{\text{nssm}}$  and bulk conductivity differently, altering their proportionality coefficient. Thus, transport properties of concrete can be compared using paste data, if aggregate type, volume, and grading are consistent.

Huang et al. [156] found a strong linear correlation ( $R^2$ =0.98) between  $D_{\rm nssm}$  and 1/F (inverse formation factor, Fig. 7), outperforming conductivity-based correlation ( $R^2$ =0.96). Since pore solution conductivity is affected by saturation, chloride migration is primarily governed by pore structure. The regression slope (2.037×10<sup>-9</sup> m²/s) aligns with chloride migration in dilute solution, confirming that 1/F reliably predicts



Materials and Structures (2025) 58:324 Page 23 of 35 324

Fable 7 The difference of D<sub>nssm</sub> procedures between concrete samples and the adapted paste tests. The last row summarizes the effect of SCM on the test results

| Procedures/setup                             | NT BUILD 492 [16]  | Huang et al. 2022 [156, 157]          | Ranger and Hasholt, 2023 [158] |
|--|--|---------------------------------------|--------------------------------|
| Curing                                       | No specific requirement  | Water curing                          | Sealed curing                  |
| Curing Time                                  | No specific requirement  | 28 d, 90 d, 180 d                     | 28 d                           |
| Pre-conditioning                             | Yes  | No                                    | Yes                            |
| Samples length                               | $50\pm2$ mm  | 50±2 mm                               | 50 mm                          |
| Sample diameter                              | 100 mm   | 50 mm                                 | 22 mm                          |
| No. of profile depth datapoints per specimen | 7  | 4-5                                   | 2                              |
| Voltage (for I <sub>initial</sub> )          | 30 V   | 40 V                                  | 30 V                           |
| Upstream solution (Catholyte vol.)           | 10 w. % NaCl in tap water  | 10 w.% NaCl (12 L)                    | 10 w. % NaCl (3.5 L)           |
| Downstream solution (Anolyte vol.)           | 0.3 M NaOH in distilled or deionised water   | 0.3 M NaOH (300 mL)                   | 0.3 M NaOH (20 mL)             |
| Temperature                                  | Maintain in the range of $20-25$ °C  | 19–20 °C                              | I                              |
| Test duration                                | 6–96 h (depends on sample)   | 24 h                                  | 24 h                           |
| SCM effects $\downarrow D_{nssm}$ mostly     | y due to refinement of pore structure by SCM (Figs. 6 & 7); linear correlation with normalised bulk conductivity | z 7); linear correlation with normali | ised bulk conductivity         |
|  |  |                                       |                                |

chloride migration in cement pastes, accounting for SCMs and hydration effects.

# 5 Final remarks and perspectives

# 5.1 Overall comparison of methods

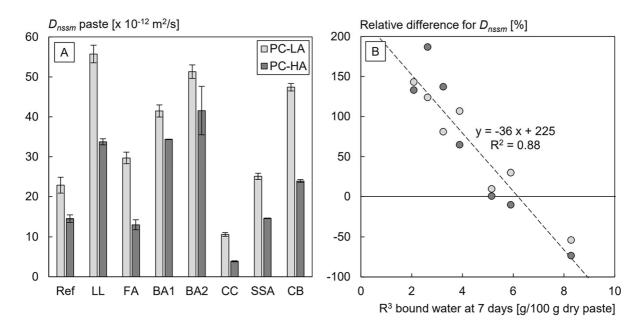
The model simplifications (with the corrections presented in Sect. 1) provide different definitions of D used in the different test methods. They are now integrated in an overview of pros and cons (Table 8), which forms a basis to summarize implications of the SCM on relevant model simplifications both from measuring and predicting point of view.

Bulk diffusion test (NT Build 492) currently appears to be the most suitable method for non-steady-state migration testing. In accelerated tests, correlations with diffusion tests are required, to use relevant *D* in model predictions. Moreover, complicated coupled processes like binding (kinetics) in accelerated tests (e.g. under and after electric fields) complicate establishing such correlations. Single-species theories neglect ion interactions, and while advanced models improve accuracy, they require refinements for practical use.

Table 9 provides a concise comparison of diffusion and migration methods for measuring chloride transport in cementitious materials, highlighting suitability for paste applications. It evaluates steady-state and non-steady-state approaches in both diffusion and migration tests, emphasizing accuracy, field relevance, and practical challenges. Diffusion and migration methods each have strengths and limitations for measuring chloride transport in paste samples. Nonsteady-state diffusion  $(D_{nssd})$  provides the most realistic and comprehensive results but requires long test durations and precise equilibrium data, while migration methods offer faster results but rely on simplifying assumptions, making them less field-relevant. Sample preparation plays a critical role in ensuring the reproducibility and reliability of chloride transport measurements in cement pastes. As outlined in Table 10, standardized guidelines—including vacuum mixing, water quality control, bleeding mitigation, and proper curing protocols—are recommended across all test types to produce uniform and representative specimens. These steps are essential to reduce artifacts such as microstructural heterogeneity or

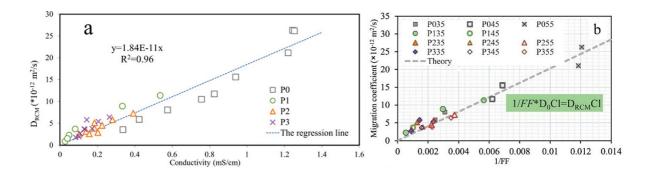


324 Page 24 of 35 Materials and Structures (2025) 58:324



**Fig. 6** Ranger and Hasholt [158]: **a** Results of miniature  $D_{\rm nssm}$  from blended cement pastes (low and high alkali PC (PC-LA and PC-HA) blended with LL-limestone; FA-Fly Ash, BA-

Biomass Ash, SSA-Sludge Ash, CB-crushed Bricks) hydrated for 28 days. **b** Effect of SCMs on Dnssm (% of the OPC value) correlated with the SCM reactivity (R<sup>3</sup> bound water test)



**Fig. 7** Huang et al. [156]: Correlation of the  $D_{\rm nssm}$  (= $D_{\rm RCM}$ ) with conductivity and **b** inverse of formation factor (1/FF [157], where formation factor is the sample conductivity nor-

malized by the conductivity of the bulk pore solution). P035 indicates he w/b, e.g. 0.35), while blended binders include FA (P1), GGBFS (P2), and LL (P3)

surface inhomogeneity, which can significantly affect transport behavior, especially in miniature-scale tests.

Building upon these preparation standards, Table 11 presents a method-specific comparison of the three primary chloride transport test types—Dssd (bulk diffusion), Dssm (steady-state migration), and Dnssm (non-steady-state migration). It provides detailed recommendations tailored to each method's operational principles, including test

durations, voltage ranges, solution types, and modeling approaches. Additionally, Table 11 highlights the practical advantages and limitations of using miniature paste-scale testing over conventional mortar or concrete-scale methods. While miniature tests offer benefits such as reduced material use, faster results, and higher sensitivity to binder composition, they may omit critical features such as aggregate-related tortuosity or realistic exposure profiles. Thus, these



 Table 8
 Comparative analysis of the models in test methods for chloride diffusivity (based on NMR or chloride measurements)

| Table 0   | Comparative analysis of the models in test inclined for choice diffusivity (based on minute of children incasarding) | decis in test incurous for cinor                        | ide dinasivity (Das | Sed on intent of children ineast  | nements)  |  |
|-----------|--|---|---------------------|---|---|--|
| Sect. No  | Test Method  | Measures  | Scale (Duration)    | Pros  | Cons  | Equation 1 corrections   |
| 3.1       | FG-NMR self-diffusion<br>(Table 3)   | Molecular mobility: $D_0 \nu s$<br>$D_{intra}$          | cm (min)            | Fast, molecular, non-invasive, no tracer, Einstein-Nernst model                   | Sophisticated equipment, in or inert porous media, chalimpurities   | Sophisticated equipment, in homogenous (ideal) solution or inert porous media, challenges with (para)magnetic impurities   |
| 3.2 & 3.3 | 3.2 & 3.3 NMR, tracer (e.g. HTO): steady state or transient  | Isotope conc. (by NMR): $D_{\rm ssd}$ or $D_{\rm nssd}$ | cm (h to days)      | Pore structure ( $D_{eff}$ ), Molecular $D_0$ and Ionic, pore $D_p$ simple models | Timely and advanced,<br>Intra- and inter-diffusion<br>in ideal solution (Table 3)   | $f_{\rm p}D_0$ (or Fick's 2nd Fleury 2019) [56]  |
| 3.1       | Steady-state diffusion,<br>(Page 1981) [6]   | CI flux: $D_{ m ssd}$                                   | cm (months)         | No binding kinetics, concentration dependency                                     | Long duration, not easy<br>to equilibrate, model<br>complexity  | $f_p f_f f_{ m E} f_p  D_0$<br>model dependent   |
| 3.3       | Bulk (transient) diffusion,<br>NT Build 443  | CI profile (Table 5): $D_{\rm nssd}$                    | cm (> 35 (90)d)     | Close to reality (under<br>seawater), total and free<br>CI should be used         | Long duration, curve fit-<br>ting without guidance,<br>challenges in free Cl<br>measurements  | $f_r f_f f_E f_p D_0 + \text{binding} + \nabla f_{\text{flux}}$  |
| 3.3       | Fick's 2nd+capillary<br>suction: Salt ponding<br>AASHTO T259, 1980<br>(2.3)  | Chloride profile (Table 5): $D_{\text{nssd+}}$          | cm (90 days)        | Two mechanisms: diffusion & capillary suction                                     | Fitting diffusion error function theory: unclear effects of<br>the two individual mechanisms, Laborious CI profile<br>measurement [72]  | on theory: unclear effects of isms, Laborious CI profile   |
| 3.3       | Fick's 2nd + water pressure [71]   | Cl profile (Table 5): $D_{\text{nssd+}}$                | cm (weeks)          | Like water permeability, sound theory   | Needs special pressure equipment, may damage specimen microstructure [160]  | oment, may damage speci-   |
| 4.1       | Steady migration NT Build<br>355, NF XP P18-461  | Downstream Cl flux: $D_{\rm sm}$                        | cm (weeks)          | Simple single species theory [139]  | long equilibration time, uncertainty in CI flux measurements due to Cl <sub>2</sub> gas formation, electrode-specimen surface $\Delta V$ drop not considered, requires multi-specic correction factor | ng equilibration time, uncertainty in Cl flux measurements due to $Cl_2$ gas formation, electrode-specimen surface $\Delta V$ drop not considered, requires multi-specie correction factor |
| 4.1       | Truc's [2000] steady migra- Upstream Cl flux: $D_{\rm ssm}$ tion [133]   | Upstream CI flux: $D_{ m ssm}$                          | cm (days)           | Shorter, overcomes many NT build 355 drawbacks                                    | Single species theory, flux may not be constant and depends on chloride binding   | nay not be constant and  |
| 4.2       | NT Build 492 or Tang's [1996] transient migration [112]  | CI penetration depth: $D_{\rm nssm}$                    | cm<br>(24–72 h)     | Shortest, simple measurement and calculation                                      | Single species theory, colorimetric method not always accurate, $\Delta V$ drop needs correction, CI binding kinet not considered   | ingle species theory, colorimetric method not always accurate, $\Delta V$ drop needs correction, CI binding kinetics not considered  |
| 4.2       | Breakthrough time transient $$ CI downstream time delay: migration $$ $$ $$ $$ $$ $$ $$ $$ $$ $$                     | CI downstream time delay: $D_{\rm nssm}$                | cm (weeks)          | Non-steady state method   | Long, theory unclear; multiple definitions of break-through time delay [161]  | ole definitions of break-  |
| 4.2       | Andrade's fitting transient migration [149]  | Cl profile (Table 5): $D_{\rm nssm}$                    | cm (day)            | Analogue diffusion theory: error function   | Laborious CI profile measurement, Diffusion theory limitations  | ement, Diffusion theory  |
| 4.2       | Equivalent time transient migration [162]  | Cl profile: D <sub>nssm</sub>                           | cm (day)            | Steady state analytical solution  | Requires correlation with natural diffusion tests   | tural diffusion tests  |
|           |  |   |                     |   |   |  |



Table 9 Comparative analysis of diffusion and migration methods with suitability for paste applications

| Category          | Test Method                                      | Pros  | Cons  | Suitability for Paste  |
|-------------------|--|---|---|--|
| Diffusion Methods | Steady-State Diffusion $(D_{\rm ssd})$           | Direct and theoretically sound measurement of $D_{\it eff}$ No transient effects thin and homogeneous paste samples   | Thin samples sensitive to<br>cracks<br>Challenging to perform<br>with large or heteroge-<br>neous samples   | Accurate for intrinsic $D_{eff}(c)$ under controlled concentration $(c)$ boundary conditions   |
|                   | Non-Steady-State Diffusion $(D_{\rm nssd})$      | Best method for chloride diffusivity measurement (realistic conditions)     can be used to study migration during hydration (no cross-section)     Provides comprehensive results for binder behavior and SCM effects | Long test duration (days to months)  Complex modeling to separate $D_{eff}$ from binding, if separate equilibrium binding input data available                            | Realistic conditions. Most<br>relevant and accurate, but<br>requires precise input for<br>binding equilibrium  |
| Migration Methods | Steady-State Migration $(D_{\rm ssm})$           | <ul> <li>Relatively quick test duration</li> <li>Measures chloride flux directly</li> <li>Applicable to damaged samples</li> <li>Applicable to paste samples</li> </ul>   | <ul> <li>Equilibration challenging</li> <li>Not field-relevant</li> <li>Assumes single species theory</li> <li>Cracks</li> </ul>  | Good for controlled<br>experiments, limited field<br>relevance;<br>linear correlation with bulk<br>conductivity  |
|                   | Non-Steady-State Migration ( $D_{\text{nssm}}$ ) | <ul><li>Relatively fast test duration (hours to days)</li><li>Applicable to concrete and paste scale</li></ul>  | <ul> <li>Mostly single species<br/>theory</li> <li>Binding equilibrium not<br/>fully accounted for</li> <li>Requires accurate chlo-<br/>ride flux measurements</li> </ul> | Suitable for rapid testing but limited by model assumptions Limited relation to $D_{eff}$ & $D_{app}$ : requires validation with $D_{nssd}$ ; Linear correlation with normalised bulk conductivity |

Table 10 Cement paste sample preparation guidelines for miniature diffusivity and migration tests

| Parameter                 | Recommendation   | Rationale/Notes   |
|---------------------------|--|---|
| Pre-mix Blended Powders   | Use two-stage water addition and vacuum mixing   | Enhances homogeneity and reduces entrapped air                    |
| Water Quality             | Use deaired, deionised or demineralised water  | Prevents impurities and air bubbles                               |
| Water-to-Binder Ratio     | Avoid high w/b ratios prone to bleeding  | Prevents segregation and ensures uniform microstructure           |
| Bleeding Control          | Rotate samples during curing; delayed mixing   | Minimizes bleeding, promotes early flocculation                   |
| Sample preparation        | Avoid core sampling; Polish or cut surface layers  | Removes inhomogeneous zones near cast surfaces                    |
| Curing Method             | Minimal water sealed curing or sacrificial slurry  | Reduces leaching while maintaining internal hydration             |
| Curing Duration           | Prefer longer curing (≥28 days) for less reactive SCMs (91 or 180 days, 1 year)                          | Promotes stable microstructure and reliable diffusion measurement |
| Coating Preparation       | Apply epoxy/polyurethane on all faces except exposure (only one basal surface coated for $D_{\rm ssd}$ ) | Ensures one-dimensional diffusion path during testing             |
| Pre-exposure conditioning | Re-saturate samples if drying occurred   | Prevents chloride ingress by advection                            |



Table 11 Recommendations for miniature diffusion and migration testing on cement pastes

|                              | or minimized windshop and magradon of  |  |   |   |
|------------------------------|--|--|---|---|
| Parameter                    | $D_{\rm ssd}$ (Bulk Diffusion)   | $D_{ m ssm}$   | $D_{ m nssm}$   | Rationale / Notes   |
| Sample size                  | Cylindrical ( $D = 33$ or 63 mm;<br>H = 10 mm) or Rectangular<br>parallelepipeds ( $150*200*15$ mm <sup>3</sup> )          | Cylindrical: $D=33$ or $63$ mm; $H=10$ mm  | Cylindrical: $D = 22 - 55$ mm;<br>$H \sim 50$ mm                                    | Ensure sample homogeneity, reduce cracking, size consistency                    |
| Pre-exposure conditioning    | Re-saturate samples if drying occurred after coating   | Re-saturation required   | Re-saturation recommended   | Prevents chloride ingress by advection (consistency)                            |
| Reservoir/ Exposure Solution | 3–3.9% NaCl (seawater),<br>~0.1 M Cl <sup>-</sup> (groundwater)  | 500 mM NaCl + 300 mM NaOH upstream; 300 mM NaOH downstream                       | 10% NaCl upstream;<br>0.3 M NaOH downstream   | Standardizes ionic environment; controls electrode reactions                    |
| Reservoir Volume / Size      | Maintain $50 \text{ cm}^2/L$ solution-to-surface ratio (ASTM C1556)  | ≥ 100 mL per cell; steady chloride gradient                                      | 3.5 or 12 L catholyte and 20 or 300 mL anolyte                                      | Controls leaching; maintains ion equilibrium                                    |
| Voltage / Current            | Not applicable   | 5–12 V (avoid overheating); ~300 V/m field strength                              | 30–40 V initial voltage; controlled to avoid damage                                 | Excessive voltage causes damage   |
| Test Duration                | up to 28 days  | 1–3 weeks (commonly ~ 14 days)   | ~24 h   | Paste downscaling accelerates testing   |
| Temperature Control          | 20–25 °C typical; maintain it stable (e.g., $20\pm2$ °C); Temperature effects identified as interesting research direction | (e.g., $20 \pm 2$ °C);<br>teresting research direction                           |   | Temperature affects ion mobility and reactions                                  |
| Coating                      | Epoxy on all but exposed face  | Epoxy coating of lateral sides   |   | unidirectional Cl <sup>-</sup> ingress  |
| Measurement Frequency        | Single (or multiple) timepoint post exposure chloride profils (three replicates)   | Frequent voltage/current<br>and ≤ daily downstream chloride<br>monitoring        | Multiple profile depth points (2 minimum; 4–5 preferred)                            | Calculation of <i>D</i> 's for pastes follow same eq. used for concrete samples |
| Chloride Profiling           | Powder titration or High-resolution (SEM–EDS, µXRF, etc.; see Table 4); Total (and 'free') chlorides                       | Optional if relying on electrical data   | Simplified profiles; fewer points suffice for paste-scale                           | Balances accuracy and test duration; challenges in free chloride detection      |
| Modeling Approach            | Square root law (Xcr/ $\sqrt{t}$ );<br>Fick's Law for $D_{\rm app}$  | Andrade's equation for effective diffusion from steady current                   | Transient migration models; time-<br>dependent migration rates                      | Each requires method-specific modelling, Single-specie theory limit             |
| Common Challenges            | Bleeding, drying shrinkage,<br>leaching  | Electrode interface degradation; solution renewal                                | Interface dissolution; current damage, CI binding uncertainty                       | Handling and electrical conditions critical; Single-specie theory limitations   |
| SCM Impact Assessment        | SCMs affect diffusion coefficient<br>and chloride binding  | 'Deff' correlates with bulk conductivity; distinguishes pore vs solution effects | $D_{ m nsm}$ correlates well with concrete; pore refinement $\downarrow D_{ m nsm}$ | Useful for SCM optimization and durability prediction                           |
|                              |  |  |   |   |



324 Page 28 of 35 Materials and Structures (2025) 58:324

| rameter D <sub>ssad</sub> (Bulk Diffusion) D <sub>ssm</sub> Rationale Lossing miniature tests) Speedup (week to month); Realise (week); Measures chloride flux comprehensive results for binder tic diffusion conditions; Provides (week); Measures chloride flux screening of SCMs' effect; small comprehensive results for binder directly; Applicable to damaged sample size allows high through behaviour and SCM effects; (paste) samples; mormalised bulk conductivity (compared to migration) tests) Paste-scale to mortar/concrete correlation with bulk conductivity and accurate and accurate and accurate compared to migration) tests) Rober from a dependent flux; binding effects (Cl-binding (require separate D <sub>eff</sub> from dependent flux; binding effects) conditions vs. real concrete cquilibrium binding tests); | ble II (continued)            |   |   |   |  |
|---|-------------------------------|---|---|---|--|
| Fastest test duration (1 day); Fast screening of SCMs' effect; small sample size allows high throughput; linear correlation with normalised bulk conductivity High voltage may distort results; Interface damage risk; Simplified conditions vs. real concrete  |                               | $D_{ m ssd}$ (Bulk Diffusion)   | $D_{ m ssm}$  | $D_{ m nssm}$   | Rationale / Notes  |
| High voltage may distort results;<br>Interface damage risk; Simplified<br>conditions vs. real concrete  | os (of using miniature tests) | Speedup (week to month); Realistic diffusion conditions; Provides comprehensive results for binder behaviour and SCM effects; Most relevant and accurate (compared to migration) tests) | Relatively fast test duration (week); Measures chloride flux directly; Applicable to damaged (paste) samples; linear correlation with bulk conductivity | Fastest test duration (1 day); Fast screening of SCMs' effect; small sample size allows high throughput; linear correlation with normalised bulk conductivity | Low material usage; Faster test duration under; Clear sensitivity to SCMs; Minimal leaching in small sealed systems; |
|   | ns (of using miniature tests) | Paste-scale to mortar/concrete correlations still missing, Complex modelling to separate $D_{\rm eff}$ from CI-binding (require separate equilibrium binding tests);                    | Artificial field conditions; single specie theory limitations; Time dependent flux; binding effects   |   | No aggregate effects; May underrepresent tortuosity and ITZ; Less realistic for service-life predictions             |

comparative insights from Table 11 support informed test selection and interpretation in research and durability assessment contexts.

# 5.2 Main findings

Based on this literature review, the following key insights can be synthesized:

- Diffusion varies by type, with ionic diffusion being the most complex. Key factors include concentration, molecular size, interactions, and ionic strength.
- PFG NMR efficiently measures self-diffusion but is limited by sensitivity to paramagnetic impurities and early hydration stages. It can eliminate binding artifacts and accurately assess pore structures.
- NMR concentration measurements, though not widely available, provide reliable insights into diffusion, porosity, and microstructure in cementitious materials.
- Traditional HTO-through diffusion methods are closest to the fundamental effective diffusion coefficients (D<sub>eff</sub>), but are time-consuming, leading to exploration of alternative methods.
- The bulk diffusion transient test protocol mimics real-world conditions (without acceleration of diffusion) and provides an engineering D<sub>app</sub> approach that lumps contributions from both microstructure and chloride binding, although not being an intrinsic property as it varies with boundary and test conditions. The common interpretation approach to fit chloride profiles with an analytical solution of Fick's second law (considering linear binding and constant surface concentration) to obtain D<sub>app</sub> is criticized due to some hypotheses and the evolution of D<sub>app</sub> with time (possibly corrected in engineering approaches with an aging factor).
- To speed up the determination of  $D_{\rm app}(t)$ , an alternative empirical approach is provided (fib Bulletin 34, 2006 [52]; Model Code 2010 [154] based on chloride migration tests  $D_{\rm nssm}$ , which should always incorporate a calibration factor to account for the correlation with  $D_{\rm app}(t)$  [68, 154].
- An alternative interpretation that has gained attention in recent years is the square root law of diffusion, which seems to describe very well the evolution of chloride penetration depths (with an offset



Materials and Structures (2025) 58:324 Page 29 of 35 324

when samples are exposed before microstructure stabilization).

- Bulk diffusion transient testing has been successfully tested at the paste scale, with the advantage of down-scaling the specimens without aggregates, allowing for greater precision and easier understanding of mechanisms. However, the casting and curing of cement pastes require special precautions because their properties are much more sensitive to inhomogeneity, bleeding, drying shrinkage, and leaching.
- Several methods allow chloride profile measurements in solid phases, each offering different levels of precision (e.g. single point analysis with colorimetry vs. profiles with < 1mm resolution vs. high resolution spectroscopic techniques) and complexity (availability of instrument, sample preparation, possibility of quantitative measurements, complexity of calibration, etc.). Variable pressure SEM-EDS seems to be a promising approach as it allows quantitative profiles with a relatively simple instrument and without extensive sample preparation.</p>

Although there is a growing interest in durability of blended cements, this review highlights that there remain many open questions on methodology how to obtain diffusion coefficients. From this study, it is concluded that the following areas of research require further investigation:

- Various models with different inputs and diffusion coefficient definitions require careful calibration and validation, integrating thermodynamics, kinetics, and microstructure to optimize cement chemistry, though separating the underlying processes remains challenging.
- Adapt NMR for materials with higher paramagnetic impurities, develop cost-effective setups, and expand its focus to inter-diffusion. Standardizing sample preparation and exploring advanced techniques for pore size and anisotropy analysis could enhance understanding of microstructural influences on diffusion.
- Extended studies on a broader range of cementitious materials (concrete, mortar, and paste) are needed to evaluate the limits of the square root diffusion law approach for profile analysis.

- A clearer distinction is needed between moderate chloride concentrations (e.g., marine environments) and high chloride concentrations (e.g., deicing salts with stronger binding effects).
- Laboratory test methods should better account for continuous leaching effects (e.g., marine exposure), as chloride ingress is influenced by hydroxyl leaching.
- Despite numerous bulk diffusion tests on cement paste, a standardized experimental protocol and repeatability studies are still lacking.
- Comprehensive bulk diffusion testing, alongside mortar and concrete assessments, is necessary to evaluate the representativeness of paste results.
- Further research is needed to quantify how SCMdriven reductions in porosity, changes in capillary pore morphology, and increases in C(A)SH density individually impact D reduction.
- The effect of SCMs on the C<sub>d</sub> value—often assumed constant in migration tests—should be reconsidered to more accurately assess D<sub>nssm</sub> in concretes with SCMs.

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Materials and Structures (2025) 58:324 Page 31 of 35 324

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324 Page 32 of 35 Materials and Structures (2025) 58:324

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