

# Research Article



CO₂ Capture

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# Amine-Functionalized Triazolate-Based Metal-Organic Frameworks for Enhanced Diluted CO<sub>2</sub> Capture Performance

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Abstract: Efficient CO<sub>2</sub> capture at concentrations between 400–2000 ppm is essential for maintaining air quality in a habitable environment and advancing carbon capture technologies. This study introduces NICS-24 (National Institute of Chemistry Structures No. 24), a Zn-oxalate 3,5-diamino-1,2,4-triazolate framework with two distinct square-shaped channels, designed to enhance CO<sub>2</sub> capture at indoor-relevant concentrations. NICS-24 exhibits a CO<sub>2</sub> uptake of 0.7 mmol/g at 2 mbar and 25 °C, significantly outperforming the compositionally related Zn-oxalate 1,2,4-triazolate – CALF-20 (0.17 mmol/g). Improved performance is attributed to amino-functions that enhance CO<sub>2</sub> binding and enable superior selectivity over N<sub>2</sub> and O<sub>2</sub>, achieving 8-fold and 30-fold improvements, respectively, in simulated CO<sub>2</sub>/N<sub>2</sub> and CO<sub>2</sub>/O<sub>2</sub> atmospheric ratios. In humid environments, NICS-24 retained structural integrity but exhibited an 85 % reduction in CO<sub>2</sub> capacity due to competitive water adsorption. Breakthrough sorption experiments, atomistic NMR analysis, and DFT calculations revealed that water preferentially adsorbs over CO<sub>2</sub> due to strong hydrogen-bonding interactions within the framework. Gained understanding of the interaction between CO<sub>2</sub> and H<sub>2</sub>O within the MOF framework could guide the modification via rational design with improved performance under real-world conditions.

#### Introduction

Indoor air quality stands as a cornerstone of public health, with profound implications for human well-being. A significant aspect of maintaining indoor air quality involves the capture and management of carbon dioxide (CO<sub>2</sub>), particularly under conditions where it accumulates in inhabited

environments due to human metabolism. This is especially critical in spaces lacking efficient ventilation systems. Elevated CO<sub>2</sub> levels, with a typical threshold concentration of 2000 ppm, can have negative impacts on human health, causing headaches and fatigue, which can escalate to nausea, dizziness, vomiting, and even death as CO<sub>2</sub> concentrations increase. Repercussions of such elevated CO<sub>2</sub> levels necessitate effective strategies for its capture and management, making this a pressing concern.

Simultaneously, the technological demand for pure CO<sub>2</sub> is increasing rapidly, yet the availability of energy-efficient processes to provide adequately pure CO<sub>2</sub> gas for various applications remains limited.<sup>[4-6]</sup> In this context, the utilization of advanced solid adsorbents, particularly through physisorption, emerges as a promising solution to tackle these intertwined challenges.<sup>[7,8]</sup>

Metal-organic frameworks (MOFs) offer great potential to mitigate these challenges due to their high flexibility in tuning the structure-property relationship by rational design. [9,10] However, the stringent requirements for indoor CO<sub>2</sub> capture, including ambient pressure and temperature conditions, selective adsorption at diluted CO2 concentrations, and performance resilience in the presence of humidity, drastically narrow down the selection of suitable MOF candidates.[11-14] The overall CO2 sorption capacity is primarily determined by the strength of CO2 sorption at low pressures and the available free pore volume at higher pressures.[15] It is already known that the most suitable pore size for CO2 storage with optimal CO2-to-MOF interaction is between 4-7 Å, and that the presence of polar functional groups and specific metal centers significantly enhance the affinity towards CO2 sorption. [16,17] Considering practical indoor air CO2 application it is imperative that efficient

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selective capture can be performed at room temperature both at higher and low pressures. MOF-74(Mg) is one of the most frequently investigated material for CO2 capture reaching the adsorption capacity of 6.3 mmol/g of adsorbent (27.5 wt %) up to 1 bar at 25°C with some other MOF systems such as UTSA-16 or UiO-66 exhibiting comparable CO<sub>2</sub> capture performances. [18-20] Precise tailoring of the pore size using of pyrazine-based ligands, along with the introduction of electrostatic SiF<sub>6</sub><sup>2-</sup> and (NbOF<sub>5</sub>)<sup>2-</sup> anions, yielded the SIFSIX and NbOFFIVE families, respectively, exhibiting single-component adsorption capacities up to 2.2 mmol of CO<sub>2</sub>/g at 400 ppm partial pressure (NbOFFIVE-1).[21,22] A new generation of Anion-Pillared MOFs (APMOFs) is also being reported for benchmark CO2 capture with customized cages. [23] A breakthrough in the design of MOFs for direct air capture technologies was achieved with the N,N'-dimethylethylene diamine (MMEN) modification of Mg<sub>2</sub>(dobpdc) (dobpdc=4,4'dioxidobiphenyl-3,3'dicarboxylate), due to cost-efficient production and achieving CO2 adsorption capacity of 3.0 mmol/g at direct air capture conditions, which is beyond the capture capacity criteria for such technologies.[24]

However, in many cases, the co-adsorption of water in a humid environment can significantly reduce the adsorption capacity for CO2, highlighting the importance of understanding and optimizing the adsorption mechanisms. [25] Notably, the presence of moisture complicates the adsorption, particularly in amino-functionalized MOFs. While competitive adsorption of water can significantly reduce the CO<sub>2</sub> capture capacity at low partial pressures, certain studies have shown that moisture can enhance CO2 uptake due to its influence on the adsorption mechanism. For instance, under dry conditions, the formation of carbamate involves two amine groups capturing one CO2 molecule. [26-28] However, under humid conditions, water can facilitate the formation of bicarbonate ions, requiring only one mole of amine per mole of CO2, thus increasing the adsorption capacity. Moreover, hydronium carbamate formation and weak hydrogen bonds between hydronium carbamate and bicarbonate have also been identified as contributing factors to enhanced CO2 capture in humid environments. [29]

Among the notable advancements, the Zn-triazolate oxalate structure (CALF-20) signifies a breakthrough in post-combustion CO<sub>2</sub> capture, owing to its facile synthesis and high capture performance even in humid conditions. Nevertheless, its sorption capacities and selectivity for CO<sub>2</sub> diminish at concentrations below approximately 5 % due to weak binding energies. Removing CO<sub>2</sub> from indoor air, however, requires efficient capture at even lower partial pressures (around 2 mbar). It has been shown that the presence of amine groups in a 3-amino-1,2,4-triazolate structure analogue (CALF-15) enhances affinity for CO<sub>2</sub> binding and promotes selective adsorption at lower partial pressures of CO<sub>2</sub>. [31]

Recognizing the potential of amino-functionalized MOFs to enhance CO<sub>2</sub>-to-framework interactions even further, our focus shifts to the design of new guanazole (3,5-diamino-1,2,4-triazole)-based oxalate MOF, designated as NICS-24. By increasing the concentration of amine groups

within the MOF framework, we aim to enhance the propensity for CO<sub>2</sub> adsorption at highly diluted concentrations, addressing a critical gap in the capabilities of existing materials like CALF-20, which is a benchmark for post-combustion CO<sub>2</sub> capture. Furthermore, we rigorously evaluate the role of water on CO<sub>2</sub> adsorption performance through breakthrough experiments and solid-state NMR experiments supported by computational approaches to gain a deeper understanding of the underlying adsorption mechanisms.

#### **Results and Discussion**

NICS-24 was synthesized using various conventional methods, each carefully designed to demonstrate the material's versatility and ease of production (experimental details in SI, Figures S1-S3). The straightforward nature of the synthesis enables scalability and the potential for fine-tuning the material's properties, being crucial for advancing from laboratory research to real-world deployment. NICS-24 structure (Zn-oxalate 3,5-diamino-1,2,4-triazolate) determined from powder XRD data (Tables S1-S4) exhibits two crystallographically distinct Zn(II) centers (Figure S4). Zn1 is in a distorted octahedral coordination environment, connected with four oxygen atoms from two oxalate anions and two nitrogen atoms from bridging diaminotriazolate ligand. Zn2 is in tetrahedral environment coordinated solely to triazole nitrogen atoms. Each type of zinc centers forms specific type of infinite chains assigned as oxalate zigzag chains for Zn1 and triazolate straight chains for Zn2 (Figure 1a and S5). Both chains are bridged with diaminotriazolate ligands resulting in generation of two types of squareshaped channels along [001] direction with the dimensions of 3.5 Å (type A) and 5.0 Å (type B) and formation of 3D framework with sqc topology (Figure 1b). The bulk product corresponds to the described structure (Figure 1c, S6).

In order to study the effect of amine functions on diluted CO<sub>2</sub> capture applications, structural features, textural properties and CO<sub>2</sub> sorption performances of NICS-24 were readily compared with compositionally related Zn(II)-1,2,4triazolate oxalate (CALF-20) and Zn(II)-3-amine-1,2,4triazolate oxalate (CALF-15) (synthesis details in Supporting Information and Figure S7). CALF-20 structure, previously elucidated by Shimizu et al., [30] comprises of Znoxalate chains linked by 1,2,4-triazolate ligands oriented in two directions. These ligands form intersecting squareshaped channels with the estimated dimensions of 4.5×4.7 Å and 4.6×5.4 Å along the [100] and [011] directions respectively. Its isostructural analogue, CALF-15, features 3amino-functionalized triazolate ligands with slightly narrower pore dimensions compared to CALF-20 (3.8×4.4 Å and 3.8×3.9 Å along [100] and [011] directions, respectively) due to the amine groups oriented inside [011] channels.[31,32] Structure schemes of investigated materials are represented in Figure S8.

All three structures exhibit consistent thermal stability, with thermogravimetric analysis (Figure S9) indicating ligand decomposition occurring within the temperature range

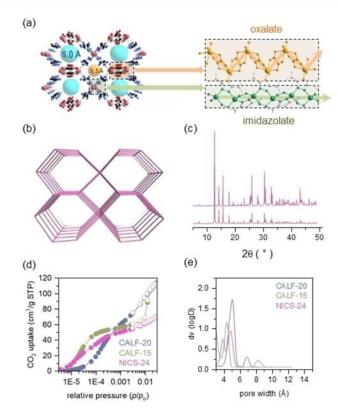


Figure 1. (a) Left - NICS-24 framework oriented along (001) direction. Spheres located within the channels symbolize free space with the indicated pore diameter estimated geometrically between opposite atoms defining the channel boundaries. Zn - purple circles, N - blue circles, C - black circles. Right - location of two type of Zn-centered chains - zigzag oxalate-based (ochre) and straight imidazolate-based (green) chain. Atoms which are not involved in Zn(II) connectivity are colored in light grey. (b) Simplified representation of NICS-24 framework indicating sqc topology. (c) XRD pattern of bulk NICS-24 sample (top) compared to the calculated pattern (below). (d) CO2 isotherms of CALF-20, CALF-15 and NICS-24 measured at 273 K (full circles adsorption points, empty circles - desorption points) with (e) correesponding NLDFT pore size distribution profiles.

of 300-400 °C. Particularly, NICS-24 maintains its framework structure with only slight change of unit cell parameters upon drying up to 250°C which shows high framework rigidity before decomposition (Figure S10, Table S5). To assess porosity properties, all three products (CALF-20, CALF-15 and NICS-24) were activated using the same twostep degassing protocol i.e. 2 h at 60 °C and 10 h at 100 °C. [30] Notably, N2 adsorption at 77 K, yielded expected BET surface area and micropore volume values solely for the CALF-20 material. Conversely, CALF-15 and NICS-24 displayed negligible N2 uptakes despite possessing comparable pore dimensions and geometries to CALF-20 (Figure S11). Lower N2 sorption capacities as expected are likely attributed to the extremely slow adsorption kinetics at cryoconditions, hindering N2 diffusion through ultra-micropore channels.[33,34] Porosity properties were therefore further evaluated using CO<sub>2</sub> sorption isotherms measured at 273 K for all samples to ensure consistency (Figures 1d, S12 and S13). CALF-20 exhibits Langmuir-type isotherm consistent with previously published data.[30] It gradually approaches saturation with a final CO<sub>2</sub> uptake of 112 cm<sup>3</sup>/g STP up to 1 bar resulting in BET surface area of 402 m<sup>2</sup>/g and a relatively broad pore size distribution with the NLDFT profile peak positioned at 0.53 nm. CALF-15 on the other hand displays Type I isotherm, reaching saturation above p/  $p_0 = 0.005$ , exhibiting pore size distribution with the peak at 0.39 nm. BET surface area is notably lower compared to CALF-20 (314 m<sup>2</sup>/g) due to narrower pores. Material demonstrates gradual gate opening effect, reflecting in significant increase of CO2 adsorption in the p/p0 pressure range between 0.01-0.015. [35] NICS-24 exhibits similar Type I isotherm as CALF-15 with slightly lower equilibrium uptake at  $p/p_0 = 0.005$  (63 cm<sup>3</sup>/g STP and 57 cm<sup>3</sup>/g STP for CALF-15 and NICS-24 respectively), resulting in a BET surface area of 274 m<sup>2</sup>/g. Bimodal pore size distribution with NLDFT profile peaks at 0.36 nm and 0.49 nm coincides with the structure possessing two types of channels (Figures 1e and S14).

Capture performance at diluted CO2 concentrations were evaluated for NICS-24 and compared with CALF-20 and CALF-15 materials using their adsorption capacity, selectivity, binding energy, regeneration capability, durability and capture performance in humidity.

Isotherms of pure gas at specific temperature and pressure regions are commonly employed as a primary screening tool to assess absolute CO2 capture capacity. However, the values obtained from such assessments do not fully reflect the true capture performance, since the working conditions such as partial pressure region, selectivity and kinetics are being neglected in that case.[36] Single-component adsorption capacity is generally influenced by accessible pore volume and BET surface area, though chemical functionality and pore environment also play crucial roles, as seen in differences between materials like activated carbons, zeolites, and MOFs. Indeed, CALF-20 with the highest surface area value among investigated materials, demonstrates the highest sorption capacity at 1 bar and 25°C, reaching 4.1 mmol/g (Figure 2a). CALF-15 and NICS-24 adsorb similar amounts of CO<sub>2</sub> (3.0 mmol/g) under these conditions. However, for specific CO2 capture application (Figure 2b), the capacity at desired partial pressures holds more relevance than the total capacity. [7,37-39] At 0.15 bar (post-combustion CO2 capture-PC), CALF-20 continues to demonstrate superiority with a CO<sub>2</sub> uptake of 2.9 mmol/g compared to CALF-15 and NICS-24, which adsorb 2.6 and 2.4 mmol of CO<sub>2</sub>/g of adsorbent, respectively. However, the sorption capacity of CALF-20 and CALF-15 decrease significantly below the 'practical threshold' at 1 mmol/g when the partial pressure of CO<sub>2</sub> is lowered to 2 mbar (indoor air capture-IAC). Conversely, NICS-24 still holds CO<sub>2</sub> uptake of 0.7 mmol/g under such conditions with the sorption capacity dropping below 1 mmol/g when the partial pressure decreases to 1 mbar (Figure 2b). The observed capacity rates NICS-24 within the broader context of amineappended MOF materials engineered for low concentration CO<sub>2</sub> capture (Table S6). Additionally, material shows high durability over 20 adsorption/desorption cycles using temperature-swing adsorption regeneration (Figure S15). Sin-

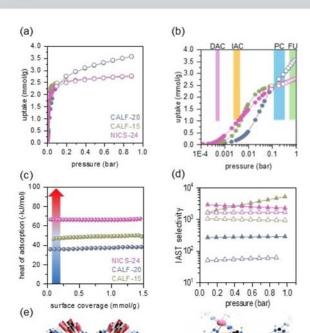


Figure 2. (a) CO2 isotherms measured at 25 °C for NICS-24, CALF-15 and CALF-20 (full circles - adsorption points, empty circles desorption points). (b) CO2 isotherms measured at 25 °C plotted in logarithmic scale. Coloured areas indicate pressure regions relevant for specific CO2 capture application; BFU - biofuel upgrade, PCU - postcombustion capture and utilization, IAC - indoor-air capture, DAC direct air capture. (c) Isosteric heat of adsorption values for CO2 for NICS-24, CALF-15 and CALF-20. Gradient-coloured arrow indicates binding type from blue - physisorption, to red - chemisorption; (d) IAST selectivity for NICS-24, CALF-15 and CALF-20. CO2/N2 at 0.2% of CO<sub>2</sub> - full symbols, CO<sub>2</sub>/O<sub>2</sub> at 0.8% of CO<sub>2</sub> - empty symbols. (e) Preferential binding sites for CO<sub>2</sub> within both types of NICS-24 channels with the indicated interatomic distances between H-atom originated from diaminotriazole amine group and O-atom from CO2 molecule (green dashed line) and calculated heat of adsorption values for specific site given within arrows.

gle-component adsorption isotherms so suggest the potential suitability of NICS-24 for diluted CO<sub>2</sub> capture applications. The reproducibility of CO<sub>2</sub> sorption performance for NICS-24 was assessed by measuring CO<sub>2</sub> isotherms across several product batches (Figure S16). The observed variability in uptake values, approximately 20% at concentrations up to 2000 ppm, demonstrates the robustness and consistency of the material's performance.

CO<sub>2</sub>-to-framework interaction energy (*Qst*) is a critical factor influencing the adsorption capacity, selectivity and the energy efficiency of regeneration processes. The optimal heat of adsorption is estimated to range between 30 and 60 kJ/mol.<sup>[40]</sup> This range balances strong binding for highly selectivity with the ability to release CO<sub>2</sub> under mild conditions. Notably, optimal range may vary with CO<sub>2</sub> concentration in the stream, with more diluted CO<sub>2</sub> requiring higher enthalpy values for effective capture and release

cycles. In this study, Qst values were calculated using Clausius-Clapeyron equation based on isotherms measured between 25 and 35 °C (Figures 2c, S17-S20 and Table S7).[41] For CALF-20, the Qst ranges from 29 and 37 kJ/mol at low surface coverages (up to 1 mmol/g), aligning with the theoretical value of 36.5 kJ/mol predicted by DFT. [30] This heat of adsorption value reflects modest physisorption, driven predominantly by van der Waals interactions. In contrast, Zn-aminotriazole oxalate (CALF-15) exhibits significantly higher Qst values between 49 and 53 kJ/mol, attributed to enhanced electrostatic interactions and spatial confinement effects within its reduced pore size. NICS-24, containing two amine groups on triazole linker, further increases the Qst to 68 kJ/mol. This value approaches the upper limit for energy-efficient regeneration particularly for temperature swing processes (TSA)[24] and highlights the role of amine-functionalized frameworks in enhancing CO2framework interactions. While the framework achieves stronger binding, the increased heat of adsorption may reflect in higher energy costs for desorption even though the interactions are still considered to be of physisorptive nature. The progression of Qst values across CALF-20, CALF-15, and NICS-24 demonstrates the interplay of pore size, amine functionality, and framework design in tailoring CO<sub>2</sub> capture performance. By synchronizing weak physisorption interactions with structural confinement effects, these materials strike varying balances between adsorption strength and regeneration efficiency.

High sorption selectivity for CO<sub>2</sub> enables delivery of the captured gas in high purity, which is pivotal for applications where CO2 utilization is targeted. Moreover, in indoor environments, selective CO2 capture prevents removal of components essential for maintaining life support systems. Sorption selectivity for CO2 against N2 and O2, the most abundant gas components in the atmosphere, was evaluated using IAST<sup>[42]</sup> considering CO<sub>2</sub> concentrations between 400 and 4000 ppm (Figures 2d and S21-S29). Since the partial pressure of O<sub>2</sub> in the atmosphere is typically four times lower than of N2, CO2/O2 ratios were adapted accordingly in the selectivity calculations. CALF-20 exhibits high CO<sub>2</sub>/N<sub>2</sub> selectivity at low CO2 concentrations, primarily due to dominant dipol-quadropole interactions between CO2 and triazolate ligands. [43] However, the presence of amine groups in CALF-15 and NICS-24 further enhances the selectivity. This is attributed to the narrower pores in CALF-15 and NICS-24 compared to CALF-20, which favor CO<sub>2</sub> adsorption over N2, and increased interactions between CO2 molecule and amine groups. A similar trend of selectivity enhancement is observed for CO<sub>2</sub>/O<sub>2</sub> (Figure 2d). Notably, lower selectivity values for oxygen compared to nitrogen in CALF-20 and CALF-15 can be attributed to the smaller kinetic diameter of O2 (3.46 Å vs 3.64 Å respectively), allowing better diffusion of O2 over N2, and the quadrupole moment of O2 which may lead to slightly stronger interactions with the framework compared to nitrogen molecule. On the other hand, CO<sub>2</sub>/N<sub>2</sub> and CO<sub>2</sub>/O<sub>2</sub> selectivity for NICS-24 are more comparable, implying that steric hindrance impedes O2 from accessing adsorption sites almost as effectively as N<sub>2</sub> in ultramicroporous channels of NICS-24.

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Theoretical calculations reveal two possible adsorption sites for CO<sub>2</sub>, in NICS-24, corresponding to two pore types, as shown in Figure 2e. The calculated adsorption energy in the B and A channels are 56 and 9 kJ/mol, respectively. These values align well with the experimentally determined heat of adsorption of 63 kJ/mol, indicating a dominant physisorption with localized interactions. In type B channel, CO<sub>2</sub> assumes a transversal position, approximately equidistant from the pore walls. It interacts weakly with the adjacent amino groups of the 3,5-diamino-1,2,4-triazolate scaffolding. The OC-O...HN(H)- distance is 2.35 Å on both sides of the pore, typical of physisorptive interactions. Bader charge analysis confirms minimal charge transfer. In isolated CO2, the charges on C and O are +1.96 e0 and -0.98 e<sub>0</sub>, respectively, which are increased to +2.00 e<sub>0</sub> and −1.00 e<sub>0</sub> when bound in NICS-24. In channel A, CO<sub>2</sub> would theoretically orient along the dimension of the pore with the OC-O...HN(H)- distance of 2.3-2.4 Å. However, due to the very low adsorption energy (-9 kJ/mol), such positioning does not occur. The upper limit of NICS-24 capacity was determined by filling both pores with CO2. Dependent on the CO2 activity, the material can uptake further CO2 if the differential adsorption energy is favorable, i.e. negative. This changes upon the addition of the fifth CO<sub>2</sub> molecule per unit cell-the first one to occupy the small pore, which puts the maximum uptake at 2.4 mmol/g which aligns perfectly with the isothermal measurements.

Most CO2 capture processes involve exposure to water vapor, a highly competitive component that can significantly impact CO2 capture efficiency. Preferential binding of water molecules to the sorption sites or ligand displacement can drastically reduce the overall sorption performance for CO2 and poses a risk of gradual degradation to the MOF structures, potentially compromising their long-term stability and functionality.[11,44]

NICS-24 exhibits high structure stability in the presence of water. Structure integrity is preserved even after soaking the material in water at 60°C for 3 days (Figure 3a, Table S8). Water isotherms measured at 25°C show substantial decrease of total water uptake for amino-functionalized structures, i.e. 11.4 mmol/g, 6.9 mmol/g and 3.4 mmol/g for CALF-20, CALF-15 and NICS-24, respectively (Figure 3b). The capacity trend is most probably related to the pore volume of the frameworks. However, isothermal step below 10% RH in the case of CALF-15 and NICS-24 indicate that the presence of amine functional groups notably contributes to the overall hydrophilicity of the amino-functionalized frameworks.

The effect of humidity on CO<sub>2</sub> capture performance was evaluated using breakthrough curve analysis. First, He gas mixture consisting of 2000 ppm CO<sub>2</sub> and 20 % O<sub>2</sub> was passed through a fixed-bed reactor at ambient temperature (Figure S30). The capacity at 25 °C in a dry environment was 0.64 mmol/g (inconsistency of CO<sub>2</sub> uptakes with isothermal data is due to the difference in sorption conditions between methods). The adsorption was followed by a He purge and increasing the temperature to 120°C to desorb physiosorbed and chemisorbed CO2, respectively. Physisorption was proved to be the dominant mechanism, since it contributes

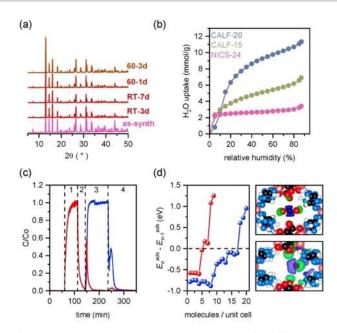


Figure 3. (a) XRD patterns of pristine NICS-24 (as-synth) and after soaking in water at room temperature for 3 days (RT-3d), 7 days (RT-7d) and at 60°C after 1 day (60-1d) and 3 days (60-3d). (b) Water adsorption isotherms of CALF-20, CALF-15 and NICS-24 measured at 25 °C. (c) Breakthrough curves showing the effect of humidity on CO2 adsorption/desorption dynamics (m/z CO2 signal - red, m/z H2O signal - blue. 1) saturating the sorbent with 2000 ppm CO2, 2) purging the system with He to remove physically bound CO2, 3) water injection to the purge gas causing CO2 desorption and 4) temperature swing desorption to remove remaining CO2. (d) Differential adsorption energy of CO2 (blue) and H2O (red) as the number of adsorbate molecules in unit cell is increased with corresponding differential electron density upon the adsorption into NICS-24 for CO2 (upper right) and H2O (lower right) (drawn with an isosurface).

57% of the total desorption capacity. With the presence of 50 %RH, the CO<sub>2</sub> capacity decreases to 0.1 mmol/g whereas the water adsorption capacity was 2.25 mmol/g. The competitive adsorption of water was additionally demonstrated by sequential adsorption experiment (Figure 3c). Upon reaching saturation of dry CO2 at 2000 ppm, physically adsorbed gas (0.29 mmol/g) was subsequently removed through an inert purge step. The sample then underwent purging with helium gas passing through the water vapor atmosphere resulting in final water uptake of 2.25 mmol/g. The water molecules apparently replace adsorbed CO2 thereby promoting the release of physisorbed CO<sub>2</sub> (0.31 mmol/g). During the following desorption step, when the temperature was increased, water and remaining strongly bonded CO2, that remained chemically bound (0.05 mmol/g), desorbed. High stability of capture performance, along with excellent structural rigidity in humid environment, was demonstrated through breakthrough cycling, which showed consistent working capacities for CO2 and H2O over ten runs. Stability was further corroborated by XRD analysis of the product after cycling protocol (Figures S31 and S32).

Preferential binding of water vs CO2 in NICS-24 was confirmed by DFT calculations. The amine groups of 3,5Research Article

diamino-1,2,4-triazolate can form hydrogen bonds with water molecules, which is reflected in stronger interaction. In the larger pore, the calculated interaction energy with H<sub>2</sub>O is 76.3 kJ/mol, while in the smaller pore, it is 44.2 kJ/ mol. In both cases, a hydrogen bond H<sub>2</sub>O···HN(H) is formed with an O-H distance of 1.99 Å. The strong interaction is maintained as up to eight H<sub>2</sub>O molecules are added within the unit cell, corresponding to a water uptake of approximately 4.4 mmol/g, which aligns well with the experimentally determined water saturation capacity (Figure 3b). Beyond this point, the differential adsorption energy gradually decays, eventually becoming slightly below 0 eV before turning positive at 17 H<sub>2</sub>O molecules per unit cell (~10 mmol/g). While the theoretical calculations suggest the capacity to accommodate up to 17 water molecules, the experimentally and energetically most favourable uptake corresponds to the first 8 H2O molecules. Larger uptake in respect to CO2 is caused by two factors: stronger interaction with H<sub>2</sub>O on account of hydrogen bonding and smaller radius of the H<sub>2</sub>O molecule as compared to CO<sub>2</sub>. The electronic effects manifest as charge transfer. Compared to isolated H<sub>2</sub>O, with charges -1.22 and +0.61 e<sub>0</sub> on O and H, respectively, the charge separation is increased to  $-1.30 e_0$ and +0.65 e<sub>0</sub>, respectively. This is visible in the differential electron density, as shown in insets of Figure 3d. The differential electron density upon adsorption confirms that CO<sub>2</sub> is still physisorbed despite relatively strong binding

To elucidate the formation of hydrogen bonds and competitive adsorption of CO<sub>2</sub> and H<sub>2</sub>O in NICS-24, we employed <sup>1</sup>H, <sup>13</sup>C, and <sup>15</sup>N MAS NMR spectroscopy. Samples were prepared ex situ with varying amounts of H<sub>2</sub>O

and CO<sub>2</sub>. Quantitative analysis of the <sup>1</sup>H and <sup>13</sup>C MAS spectra was used to determine the precise molar amount of H<sub>2</sub>O/CO<sub>2</sub> per unit cell. The <sup>1</sup>H MAS spectra exhibit a broad signal at 4.6 ppm attributed to the protons of NH<sub>2</sub> functional groups, and a narrow signal between 2-3 ppm corresponding to water protons (Figure 4a). The <sup>13</sup>C MAS NMR confirms the expected 1:2 ratio between oxalate carbon nuclei (around 171 ppm) and guanazolate carbon nuclei (163 ppm) and reveals an additional peak at 124 ppm assigned to encapsulated CO2 (Figure S33). Strong spinning sidebands of the CO2 peak at slower MAS rates indicate restricted dynamics of CO<sub>2</sub>, suggesting binding within the micropores of NICS-24. The 1H-13C CP-HETCOR spectrum shows a cross-peak between CO2 and NH2 resonating at slightly shifted <sup>1</sup>H frequency (4.8 ppm), indicating deshielding of the NH<sub>2</sub> proton due to hydrogen bonding with CO<sub>2</sub> (Figure S34). The restricted dynamics of hydrogen-bonded NH<sub>2</sub> prevent averaging of the chemical shifts between the two inequivalent protons attached to the same nitrogen atom. The unbound NH<sub>2</sub> proton resonates at ~4.4 ppm and is also in close proximity to the CO2 carbon, suggesting that the CO2 molecule is slightly tilted relative to the axis of symmetry of the NH<sub>2</sub> group. Surprisingly, CO<sub>2</sub> also interacts with H<sub>2</sub>O, giving rise to another correlation peak in the <sup>1</sup>H-<sup>13</sup>C CP-HETCOR at the <sup>1</sup>H position of 2.6 ppm. This signal is shifted by ~0.6 ppm downfield compared to the sample with the same amount of water but without CO2, confirming the OCO...HO hydrogen bond formation. <sup>1</sup>H-<sup>13</sup>C CP-HETCOR also supports the formation of hydrogen bond between oxalate oxygen and NH2 of guanazolate linker, which is consistent with the proposed structure and DFT calculations.

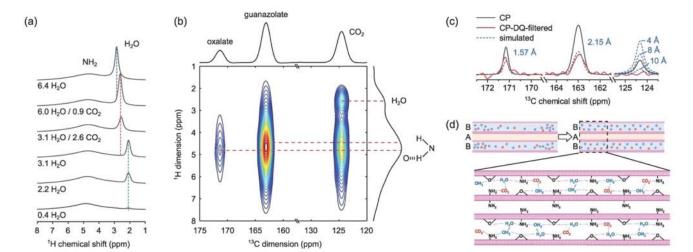


Figure 4. (a) <sup>1</sup>H MAS NMR spectra of NICS-24 with varying CO<sub>2</sub>/H<sub>2</sub>O compositions. Dashed lines highlight three distinct local environments of water molecules, distinguished by different numbers and/or H-bond strengths: partially-hydrated (green), fully-hydrated (blue), and CO<sub>2</sub>-water mixed phases (red). (b) <sup>1</sup>H-<sup>13</sup>C CP-HETCOR NMR spectrum of NICS-24 with 6 H<sub>2</sub>O and 0.9 CO<sub>2</sub> molecules per unit cell. Dashed lines indicate <sup>1</sup>H chemical shifts of water and two NH<sub>2</sub> protons, with one slightly donwnfield shifted due to H-bonding, (c) <sup>1</sup>H-<sup>13</sup>C CP-MAS (black solid) and <sup>13</sup>C CP-DQ-filtered (red solid) NMR spectra of NICS-24 containing 6 H<sub>2</sub>O and 0.9 CO<sub>2</sub> molecules per unit cell, alongside simulated 13 C CP-DQ-filtered (blue dashed) peaks for oxalates (C–C distances of 1.57 Å), guanazolates (C–C distance of 2.15 Å), and CO<sub>2</sub> molecule with C–C distances of 4 Å, 8 Å and 10 Å, assuming 99% <sup>13</sup>C-enrichment of CO<sub>2</sub>. (d) Schematic representation of CO<sub>2</sub>/H<sub>2</sub>O adsorption mechanism, illustrating the initial clustering of adsorbate molecules close to the entrance of B-type channels, whereas A-type channels remain empty (upper left), and the tendency for homogeneous distribution and formation of CO<sub>2</sub>-H<sub>2</sub>O pairs at the equilibrium (upper right and down).

Amine groups are known for their hydrophilicity and the <sup>1</sup>H-<sup>1</sup>H SQ-SQ homonuclear-correlation NMR experiment indeed confirms proximities between H<sub>2</sub>O and NH<sub>2</sub> species (Figure S35a), and <sup>1</sup>H-<sup>1</sup>H DQ-SQ, which is more sensitive to shorter distances and can in addition to SQ-SQ detect correlations between equivalent protons, further confirms the existence of two distinct proton sites in NH2 moieties, with one slightly shifted due to hydrogen bond formation (Figure S35b). Moreover, in 1H-15N CPMAS spectrum we can also resolve two NH2 sites as evident from Figure S35b, but 1H-15N CP-HETCOR clearly shows these two sites are both involved in hydrogen bonding to some extent as there is hardly any difference in the center of the gravity for the two cross-peaks along <sup>1</sup>H dimension (Figure S36). PXRD analysis revealed that half of NH2 groups point towards B channels and the other half towards A channels, experiencing slightly different local environments, resulting in peak splitting. The rest of the 15N peaks of guanazolate linkers were not split but were still influenced by the content inside the large pores due to the nearby NH2 groups involved in host-guest hydrogen bonding. Conversely, chemical shift of NH<sub>2</sub> within B channels seems less affected by guest molecules, suggesting no molecules enter the smaller channels. Similarly, the <sup>13</sup>C peak position of oxalates shifted with increasing water content, further confirming interaction of oxalates with H<sub>2</sub>O inside the large pores (Figure S36). With solid-state NMR we were able to follow different sorption sites. 1H MAS NMR showed that with up to four H<sub>2</sub>O molecules per unit cell, water molecules reside in magnetically equivalent local environments, forming weak NH--OH and OH-OCC hydrogen bonds, resulting in slight deshielding of the water protons from the expected 1.5 ppm for isolated water molecules, shifting to about 2 ppm in NICS-24. When more than four H<sub>2</sub>O molecules per unit cell are introduced, additional or slightly stronger hydrogen bonds are formed (most likely between the neighboring H<sub>2</sub>O), shifting the peak to a final position of 2.85 ppm. Unlike in case of NH<sub>2</sub>, the diffusion of water and/or proton hopping prevents the observation of distinct water protons; however, the averaged chemical shift serves as a good indicator of the hydrogen-bonded network strength.

In NICS-24, the interplay between CO2 and H2O molecules is inevitable. The total number of O atoms per unit cell remains constant at 8, indicating that every molecule of CO<sub>2</sub> is replaced by two water molecules after air exposure, forming a slightly stronger hydrogen-bonded network (2.85 ppm vs. 2.6 ppm). But even when the material is partially filled with water (signal at 2 ppm), it continues to strongly adsorb either CO<sub>2</sub> or preferentially H<sub>2</sub>O. Furthermore, CO<sub>2</sub> has a tendency to evenly distribute along the channels without clustering, as no CO2-CO2 pairs were detected at C-C distances up to 10 Å, evident from the 13C CP-DQ-filtered experiment (Figure 4c). Interestingly, this tendency of forming H2O-CO2 pairs is stronger than preserving fully-water occupied unit cells in the equilibrium. A phase segregation was observed initially when H<sub>2</sub>O was introduced into CO2-loaded NICS-24, with separate H2O-H<sub>2</sub>O and H<sub>2</sub>O-CO<sub>2</sub> domains (Figure S35). After several days of the sample in a closed rotor, CO2 redistributed uniformly

within the framework, as evidenced by a single <sup>1</sup>H peak at ~2.6 ppm. This redistribution (shown in Figure 4d) illustrates the framework's ability to dynamically adjust to varying guest molecule ratios. After the rotor was opened and exposed to air, CO<sub>2</sub> exchanged with water vapor, causing a shift of the <sup>1</sup>H signal to 2.85 ppm, highlighting the material's preference for water over CO2 adsorption under ambient conditions.

In summary, the study revealed that the amine groups within NICS-24 enhance CO2 interactions through specific hydrogen bonding, which is modulated by the presence of water. Initially, CO2 and water cluster near type B channel entrances and are then redistributed over time, achieving a homogenized equilibrium state characterized by specific interactions with the framework sorption sites and between each other (Figure 4d). Unlike the commonly reported amine-functionalized MOF adsorbents, NICS-24 does not exhibit cooperative CO2-H2O adsorption via hydronium carbamate formation. This behaviour is attributed to its structural and chemical properties: (a) the amine groups in NICS-24 are not spatially aligned to stabilize such species, as cooperative adsorption requires closely positioned amines for carbamate formation; (b) its strong hydrophilicity, confirmed by NMR and computational analyses, favours water competing for adsorption sites, limiting CO2 binding efficiency, (c) its ultra-microporous channels impose steric constraints, preventing the formation of bulkier species. These findings emphasize the balance between competition and synergy in NICS-24, driven by its framework properties, and highlight its capability for CO2 capture under humid conditions despite efficiency trade-offs.

#### Conclusion

This study establishes NICS-24, a diaminotriazole oxalate Zn(II)-based MOF, as a promising material for low-concentration CO<sub>2</sub> capture, particularly in indoor environments. Through the incorporation of amine functionalities, NICS-24 exhibited significant enhancements in CO<sub>2</sub> adsorption, achieving an 8-fold increase in uptake compared to the nonamine functionalized CALF-20 and a 40% improvement over the mono-amino CALF-15 at indoor-relevant conditions (2000 ppm CO<sub>2</sub> at 25 °C). These findings demonstrate the efficacy of tailoring framework chemistry to strengthen CO<sub>2</sub>-framework interactions, enabling effective adsorption at low partial pressures.

Enhanced CO2 binding comes at the cost of increased hydrophilicity, leading to competitive water adsorption in humid conditions. Multi-aspect approach which included dynamic breakthrough experiments, supported with computational DFT and experimental atomistic solid-state NMR revealed that the strong hydrogen bonding between water molecules and the amine groups in NICS-24 diminishes CO2 uptake by occupying key adsorption sites. This trade-off between CO2 affinity and water resistance underscores the challenges in designing MOFs for realistic, moisture-rich environments. Despite this limitation, the structural integrity



of NICS-24 remained intact, indicating its robustness and adaptability for further optimization.

The study highlights a critical balance that must be achieved in the design of MOFs for  $\mathrm{CO}_2$  capture: while strong binding energies are desirable for capturing low-concentration  $\mathrm{CO}_2$ , they can also lead to challenges in humid environments where water competes for the same adsorption sites. This trade-off between binding energy strength and efficient  $\mathrm{CO}_2$  capture in diluted, wet conditions underscores the need for fine-tuning the material's properties by a proper structural modification or composite material to mitigate water interference while maintaining high  $\mathrm{CO}_2$  capture performance.

The authors have cited additional references within the Supporting Information.  $^{[24,30,31,35,45-74]}$ 

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## Conflict of Interest

The authors declare no conflict of interest.

#### **Data Availability Statement**

The data that support the findings of this study are available from the corresponding author upon reasonable request.

**Keywords:** diluted  $CO_2$  capture  $\cdot$  amino-functionalized MOFs  $\cdot$  wet  $CO_2$  adsorption  $\cdot$  indoor air purification  $\cdot$  solid-state NMR analysis

- [1] K. W. Tham, Energy Build. 2016, 130, 637.
- [2] S. A. Ahmed Abdul-Wahab, S. C. F. En, A. Elkamel, L. Ahmadi, K. Yetilmezsoy, Atmos. Pollut. Res. 2015, 6, 751–767.
- [3] D. P. Wyon, Indoor Air 2004, 14 (7), 92.
- [4] C. Kim, C.-J. Yoo, H.-S. Oh, B. K. Min, U. Lee, J. CO2 Util. 2022, 65, 102239.
- [5] R. S. Norhasyima, T. M. I. Mahlia, J. CO2 Util. 2018, 26, 323.
- [6] S. Valluri, V. Claremboux, S. Kawatra, J. Environ. Sci. 2022, 113, 322.
- [7] L. Jiang, W. Liu, R. Q. Wang, A. Gonzalez-Diaz, M. F. Rojas-Michaga, S. Michailos, M. Pourkashanian, X. J. Zhang, C. Font-Palma, *Prog. Energy Combust. Sci.* 2023, 95, 101069.
- [8] F. Zerobin, T. Pröll, Ind. Eng. Chem. Res. 2020, 59, 9207.
- [9] Z. Chen, K. O. Kirlikovali, L. Shi, O. K. Farha, *Mater. Horiz.* 2023, 10, 3257.
- [10] Z. Ji, H. Wang, S. Canossa, S. Wuttke, O. M. Yaghi, Adv. Funct. Mater. 2020, 30, 2000238.
- [11] J. Canivet, A. Fateeva, Y. Guo, B. Coasne, D. Farrusseng, Chem. Soc. Rev. 2014, 43, 5594.

- [12] J. An, O. K. Farha, J. T. Hupp, E. Pohl, J. I. Yeh, N. L. Rosi, Nat. Commun. 2012, 3, 604.
- [13] G. Centi, S. Perathoner, Microporous Mesoporous Mater. 2008, 107. 3.
- [14] S. Mahajan, M. Lahtinen, J. Environ. Chem. Eng. 2022, 10, 108930.
- [15] R. Babarao, J. Jiang, Langmuir 2008, 24, 6270.
- [16] Q. Yang, C. Zhong, J. F. Chen, J. Phys. Chem. C 2008, 112, 1562.
- [17] A. Torrisi, R. G. Bell, C. Mellot-Draznieks, Cryst. Growth Des. 2010, 10, 2839.
- [18] C. A. Trickett, A. Helal, B. A. Al-Maythalony, Z. H. Yamani, K. E. Cordova, O. M. Yaghi, Nat. Rev. Mater. 2017, 2 (8), 17045.
- [19] A. Masala, J. G. Vitillo, G. Mondino, C. A. Grande, R. Blom, M. Manzoli, M. Marshall, S. Bordiga, ACS Appl. Mater. Interfaces 2017, 9 (1), 455–463.
- [20] A. Koutsianos, E. Kazimierska, A. R. Barron, M. Taddei, E. A. Andreoli, *Dalton Trans.* 2019, 48, 3349.
- [21] O. Shekhah, Y. Belmabkhout, Z. Chen, V. Guillerm, A. Cairns, K. Adil, M. Eddaoudi, Nat. Comm. 2014, 5, 4228.
- [22] P. M. Bhatt, Y. Belmabkhout, A. Cadiau, K. Adil, O. Shekhah, A. Shkurenko, L. J. Barbour, M. Eddaoudi, J. Am. Chem. Soc. 2016, 138 (29), 9301–9307.
- [23] Y. Hu, Y. Jiang, J. Li, L. Wang, M. Steiner, R. F. Neumann, B. Luan, Y. Zhang, Adv. Funct. Mater. 2023, 33, 2213915.
- [24] T. M. McDonald, W. R. Lee, J. A. Mason, B. M. Wiers, C. Seop Hong, J. R. Long, J. Am. Chem. Soc. 2012, 134, 7056–7065.
- [25] J. M. Kolle, M. Fayaz, A. Sayari, Chem. Rev. 2021, 121, 7280.
- [26] W. R. Lee, S. Y. Hwang, D. W. Ryu, K. S. Lim, S. S. Han, D. Moon, J. Choi, C. S. Hong, Energy Environ. Sci. 2014, 7, 744.
- [27] H. Dong, L.-H. Li, Z. Feng, Q.-N. Wang, P. Luan, J. Li, C. Li, ACS Materials Lett. 2023, 5, 2656.
- [28] R. Ben Said, J. M. Kolle, K. Essalah, B. Tangour, A. Sayari, ACS Omega 2020, 5, 26125.
- [29] D. D. Miller, J. Yu, S. S. C. Chuang, J. Phys. Chem. C 2020, 124, 24677.
- [30] J.-B. Lin, T. T. T. Nguyen, R. Vaidhyanathan, J. Burner, J. M. Taylor, H. Durekova, F. Akhtar, R. K. Mah, O. Ghaffari-Nik, S. Marx, N. Fylstra, S. S. Iremonger, K. W. Dawson, P. Sarkar, P. Hovington, A. Rajendran, T. K. Woo, G. K. H. Shimizu, Science 2021, 374, 1464.
- [31] R. Vaidhyanathan, S. S. Iremonger, K. W. Dawson, G. K. H. Shimizu, Chem. Commun. 2009, 35, 5230.
- [32] R. Vaidhyanathan, S. S. Iremonger, G. K. H. Shimizu, P. G. Boyd, S. Alavi, T. K. Woo, *Science* 2010, 330, 650.
- [33] K. C. Kim, T.-U. Yoon, Y.-S. Bae, Microporous Mesoporous Mater. 2016, 224, 294–301.
- [34] Y. Hu, Y. Chen, W. Yang, J. Hu, X. Li, L. Wang, Y. Zhang, Sep. Purif. Technol. 2024, 343, 127099.
- [35] A. Banerjee, S. Nandi, P. Nasa, R. Vaidhyanathan, Chem. Commun. 2016, 52 (9), 1851.
- [36] Z. Hu, Y. Wang, B. B. Shah, D. Zhao, Adv. Sustainable Syst. 2018, 3, 1800080.
- [37] J. A. Garcia, M. Villen-Guzman, J. M. Rodriguez-Maroto, J. M. Paz-Garcia, J. Environ. Chem. Eng. 2022, 10, 108470.
- [38] F. Raganati, F. Miccio, P. Ammendola, Energy Fuels 2021, 35, 12845.
- [39] L. R. López, P. Dessi, A. Cabrera-Codony, L. Rocha-Melogno, N. J. R. Kraakman, M. D. Balaguer, S. Puig, Clean. Eng. Technol. 2024, 20, 100746.
- [40] Y. Belmabkhout, V. Guillerm, M. Eddaoudi, Chem. Eng. J. 2016, 296, 386.
- [41] A. Nuhnen, C. Janiak, Dalton Trans. 2020, 49, 10295.
- [42] K. S. Walton, D. S. Sholl, AIChE J. 2015, 61, 2757.

521373, 2025, 14, Downloaded from https://onlinelbrary.wiley.com/doi/10.1002/anie.202424747 by Kemijski Institut, Wiley Online Library on [08/04/2025]. See the Terms and Conditions (https://onlinelbrary.wiley.com/terms-and-conditions) on Wiley Online Library for rules of use; OA articles are governed by the applicable Creative Commonsors.

[43] K. Gopalsamy, D. Fan, S. Naskar, Y. Magnin, G. Maurin, ACS

Appl. Energ. Mater. 2024, 2, 96.

- [44] E. González-Zamora, I. A. Ibarra, Mater. Chem. Front. 2017, 1, 1471.
- [45] B. Grünberg, T. Emmler, E. Gedat, I. Shenderovich, G. H. Findenegg, H. Limbach, G. Buntkowsky, *Chem. A Eur. J.* 2024, 10, 5689.
- [46] C. Dalvit, M. Veronesi, A. Vulpetti, J. Magn. Reson. Open 2022, 12–13, 100070.
- [47] M. N. C. Zarycz, C. Fonseca Guerra, J. Phys. Chem. Lett. 2018, 9, 3720.
- [48] J. Peng, H. Wang, D. H. Olson, Z. Li, J. Li, Chem. Commun. 2017, 53, 9332.
- [49] R. Zhang, Y. Nishiyama, P. Sun, A. Ramamoorthy, J. Magn. Reson. 2015, 252, 55.
- [50] M. Feike, D. E. Demco, R. Graf, J. Gottwald, S. Hafner, H. W. Spiess, J. Magn. Reson. Ser. A 1996, 122, 214.
- [51] A. Altomare, C. Cuocci, C. Giacovazzo, A. Moliterni, R. Rizzi, N. Corriero, A. Falcicchio, J. Appl. Crystallogr. 2013, 46, 1231.
- [52] Bruker AXS (2018) TOPAS, v 6.0.
- [53] G. Metz, X. L. Wu, S. O. Smith, J. Magn. Reson. Ser. A 1994, 110, 219.
- [54] A. E. Bennett, C. M. Rienstra, M. Auger, K. V. Lakshmi, R. G. Griffin, J. Chem. Phys. 1995, 103, 6951.
- [55] I. Scholz, P. Hodgkinson, B. H. Meier, M. Ernst, J. Chem. Phys. 2009, 130, 114510.
- [56] M. Kotecha, N. P. Wickramasinghe, Y. Ishii, Magn. Reson. Chem. 2007, 45, S221.
- [57] M. Hohwy, H. J. Jakobsen, M. E. Edén, M. H. Levitt, N. C. Nielsen, J. Chem. Phys. 1998, 108, 2686.
- [58] M. Bak, J. T. Rasmussen, N. C. Nielsen, J. Magn. Reson. 2000, 147, 296.
- [59] A. E. Bennett, C. M. Rienstra, J. M. Griffiths, W. Zhen, P. T. Lansbury, R. G. Griffin, J. Chem. Phys. 1998, 108, 9463.
- [60] M. Shen, B. Hu, O. Lafon, J. Trébosc, Q. Chen, J.-P. Amoureux, J. Magn. Reson. 2012, 223, 107.

- [61] S. Choi, T. Watanabe, T. H. Bae, D. S. Sholl, C. W. Jones, J. Phys. Chem. Lett. 2012, 3, 1136–1141.
- [62] W. R. Lee, S. Y. Hwang, D. W. Ryu, K. S. Lim, S. S. Han, D. Moon, J. Choi, C. S. Hong, *Energy Environ. Sci.* 2014, 7, 744–751.
- [63] H. Jo, W. R. Lee, N. W. Kim, H. Jung, K. S. Lim, J. E. Kim, D. W. Kang, H. Lee, V. Hiremath, J. G. Seo, H. Jin, D. Moon, S. S. Han, C. S. Hong, *ChemSusChem* 2017, 10, 541–550.
- [64] W. R. Lee, J. E. Kim, S. J. Lee, M. Kang, D. W. Kang, H. Y. Lee, V. Hiremath, J. G. Seo, H. Jin, D. Moon, M. Cho, Y. Jung, C. S. Hong, *ChemSusChem* 2018, 11, 1694–1707.
- [65] J. Park, J. R. Park, J. H. Choe, S. Kim, M. Kang, D. W. Kang, J. Y. Kim, Y. W. Jeong, C. S. Hong, ACS Appl. Mater. Interfaces 2020, 12, 50534–50540.
- [66] L. A. Darunte, A. D. Oetomo, K. S. Walton, D. S. Sholl, C. W. Jones, ACS Sustainable Chem. Eng. 2016, 4, 5761–5768.
- [67] H. Li, K. Wang, D. Feng, Y. P. Chen, W. Verdegaal, H. C. Zhou, ChemSusChem 2016, 9, 2832–2840.
- [68] D. Jo, S. K. Lee, K. H. Cho, J. W. Yoon, U. H. Lee, ACS Appl. Mater. Interfaces 2022, 14, 56707–56714.
- [69] Z. Shi, Y. Tao, J. Wu, C. Zhang, H. He, L. Long, Y. Lee, T. Li, Y. B. Zhang, J. Am. Chem. Soc. 2020, 142, 2750–2754.
- [70] O. T. Qazvini, S. G. Telfer, ACS Appl. Mater. Interfaces 2021, 13, 12141–12148.
- [71] O. T. Qazvini, S. G. Telfer, J. Mater. Chem. A 2020, 8, 12028– 12034.
- [72] J. An, S. J. Geib, N. L. Rosi, J. Am. Chem. Soc. 2010, 132, 38-
- [73] S. Couck, J. F. M. Denayer, G. V. Baron, T. Rémy, J. Gascon, F. Kapteijn, J. Am. Chem. Soc. 2009, 131, 6326–6327.
- [74] G. E. Cmarik, M. Kim, S. M. Cohen, K. S. Walton, *Langmuir* 2012, 28, 15606–15613.

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