

Permanent Hydrophobic Coating of Chitosan/Cellulose nanocrystals Composite Film by Cold Plasma Processing

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Abstract

A highly effective ultrafast chemical modification of chitosan/cellulose nanocrystals composite film surface is presented in the present research. Added value properties were achieved by the use of a few second's treatment with RF-generated low-temperature fluorocarbon plasma. Drastic increase of the water-repelling character of the completely natural-based biocomposite foil were observed, with contact angle reaching up to 121°. Surface fluorination occurred through formation of irreversible fluorine-related bonds (CF-CF₂, CF₂-CF₂ and C-CF, CF₃ and O-CF₂, O-CF₃) detected by the means of X-ray photoelectron spectroscopy. Surface structural changes were fur-

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ther confirmed with ATR-FTIR. With packaging application in mind, the films were subjected to analysis of mechanical and water-related properties, showing an improvement upon fluorination. Further the stability of the modification was followed by measurements of water contact angle and atomic composition, as well as mechanical properties, water content and water vapour transmission after at least 31 days of storage in controlled environment. Lastly, no leaching of fluorinated components into liquid environments was detected by the means of HILIC LC-MS analyses in ESI(+) and ESI(-) modes. The presented method rapidly enhances the hydrophobic character of chitosan/nanocellulose biocomposite films without receding mechanical strength and provides a long-lasting surface coating.

Keywords: Chitosan/Cellulose nanocrystals film, Biocomposite packaging, Functionalization with cold plasma processing, Permanent hydrophobic Coating, Shelf-life of modification

1. Introduction

An alarming amount of plastic packaging is produced and discarded every year. Worldwide manufacture of plastic reached 368 million tonnes in 2019, with packaging sector taking up 39.6 % of overall demand [1]. Plastic intended for packaging also has the shortest life cycle with average of only one year and includes non-recyclable plastics such as LDPE and PS [1, 2]. Additionally, the conventional plastic is based on non-renewable resources, which emphasizes the need for an alternative even further.

Chitosan, obtained by deacetylation of chitin, that is found in food-processing industry waste, is one of the promising sustainable materials of the future [3]. Owing to its film-forming ability, focus on chitosan-based films as a potential packaging material is emerging. Besides being non-toxic, bio-compatible and biodegradable, these films exhibit not only sufficient mechanical strength and elasticity, but also antioxidant and antimicrobial properties [4, 3, 5]. Furthermore, the desired characteristics can be tuned by incorporation of various additives, namely plant extracts [6, 7, 8], essential oils [9, 10, 11] into chitosan matrix and was proven to extend the shelf life of various fresh foods [12, 13]. Mechanical properties can be improved by incorporation of cellulose nanocrystals, rod-like shaped nanoparticles isolated from cellulose, the most abundant polymer on earth [14].

Yet, one of the crucial roles of packaging is to protect its content from environmental factors, hence water resistance and low water vapor transmission are desired. Chitosan film exhibits poor moisture barrier, which limits

24 its wider use [15]. For reference, LDPE, PVC and PLA transmit 1.5 g m^{-2} ,
25 3 g m^{-2} and 40 g m^{-2} of water vapor daily, respectively, while chitosan wa-
26 ter vapor transmission rate is as much as $150 \text{ g m}^{-2} \text{ day}^{-1}$ [16]. To tackle
27 this drawback, two approaches of film modification have been researched: i-
28 introducing hydrophobic components such as fatty acids into a film-forming
29 solution [17, 18, 19], or ii- chemically modify polysaccharide [17], while modi-
30 fication of the bio-polymer itself before obtaining a film-forming solution was
31 generally shown not to be suitable, as these derivatives often do not exhibit
32 film-forming properties [17]. The first described path to hydrophobization
33 of chitosan leads to composite materials derived from emulsion which can
34 negatively influence other properties, or to multilayer materials [17, 18]. For
35 chemical modification of chitosan surface, various polymer grafting paths
36 were proposed. Up to date literature describes grafting of N-acetyl cys-
37 teine to chitosan (resulted in hydrated films with improved tensile properties)
38 [20], alkyl chains to chitosan with intention to improve water resistance for
39 bonding applications [21] and grafting of poly(2-hydroxyethyl methacrylate)
40 chains on chitosan followed by esterification with a fluorinated compound,
41 which is responsible for hydrophobicity [15]. In the latter, the rise in con-
42 tact angle was 16° and was stable through 10 min of wetting. The drawback
43 of such modification is that the process involves several steps, can be quite
44 time-consuming and produce liquid chemical waste. On the contrary, plasma
45 processing is a single-step modification, eliminating subsequent separation
46 techniques and producing no liquid discard. RF-generated plasma is a mul-

47 tifunctional tool for etching of the surface or plasma-enhanced chemical vapor
48 deposition, with thermal energy of the electrons being few eV, bringing the
49 atoms to excited state and inducing chemical reaction. Still, the temperature
50 of the electron gas remains low (near room-temperature), allowing modifica-
51 tions of sensitive surfaces, such as biopolymers [22, 23]. The closed reactor
52 system offers the possibility of gas capturing and recycling, preventing harm-
53 ful gas exhaust into the environment [24, 25]. Furthermore, the possibility
54 of scale-up and utilization of the same assembly system for tailoring the ma-
55 terials surface properties with various dopants (nitrogen, oxygen, sulfur, or
56 fluorine) highlights the industrial applicability of the technique [26, 27, 28].

57 Despite simplicity, effectiveness and rapidity of plasma treatment, only
58 limited amount of studies concerning chitosan surface functionalization are
59 taking this approach. Several researchers treated chitosan with argon plasma
60 yielding a more hydrophilic surface required in adhesives and coatings, but is
61 not desired for packaging [29, 30, 31, 19]. Slight increase in hydrophobicity,
62 water contact angle rose from initial 13° to 23° , was achieved with alkane
63 vapor plasma technique [32]. On the example of cellulose textiles and films,
64 it has already been demonstrated that plasma using fluorinated compounds
65 as a carrier gas successfully modifies surface and enhances its hydrophobic
66 character [33, 34, 35, 36, 28].

67 Chow et al. [37] reported general non-cytotoxicity of fluorinated chitin
68 derivates, leading to increased interest in use of these materials as wound
69 dressing agents or topical medication, taking in advantage their antibacte-

70 rial activity [38, 39]. Furthermore, as demonstrated on example of fluorine-
71 modified cellulose [40], fluorinated biopolymers can be enzymatically de-
72 graded by dehalogenases in several microorganisms, that are present in soil
73 and aqueous environment [41, 42].

74 With respect to these facts, the present study focuses on utilization
75 of RF-generated plasma in fluorocarbon (CF_4) for improvement of water-
76 related properties of chitosan-based films reinforced with cellulose nanocrys-
77 tals (CNCs) as a fast, efficient, non-destructive and stable modification pro-
78 cess, to best of our knowledge, for the first time. The samples were treated
79 for 5 s, which was the optimal treatment time as determined according to
80 water contact angle measurement, and subjected to surface analyses with
81 XPS, FTIR-ATR and SEM. Further, physicochemical properties relevant for
82 packaging application were considered (mechanical properties, moisture con-
83 tent and water vapor transmission. To prove stability of modification, which
84 can be deemed as uncertain, the water contact angle was followed through a
85 period of 40 days, while other properties were re-evaluated after 31 days of
86 storage in controlled environment. Finally, the hydrophobized films stability
87 and possible migration of fluorinated species into liquid environments was
88 examined, to validate applicability of plasma-treated chitosan-based films as
89 packaging.

90 2. Experimental Section

91 2.1. Materials

92 High molecular weight chitosan (85 % deacetylated), lactic acid (85 %
93 aqueous solution) and ammonium formate (for LC-MS) were bought from
94 Sigma-Aldrich (Steinheim, Germany). Cellulose nanocrystals were supplied
95 by Navitas (Stari trg pri Ložu, Slovenia). Glycerol was acquired from Phar-
96 machem Sušnik (Ljubljana, Slovenia), while ethanol (absolute), acetic acid
97 (glacial 100 % and 100 %, for LC-MS) and ammonia solution (32 %) were
98 purchased from Merck (Darmstadt, Germany). Acetonitrile for LC-MS was
99 obtained from Honeywell (North Carolina, USA) and ultrapure water (18
100 $\text{M}\Omega^{-1} \text{ cm}$) was supplied by a Milli-Q water purification system (Millipore,
101 Bedford, MA, USA).

102 2.2. Fabrication of Chitosan-Based Films with Incorporated CNCs

103 Chitosan-based films were prepared according to a previously described
104 protocol [14]. Detailed description can be found in Appendix A.

105 2.3. Hydrophobization of Films by Fluorocarbon (CF_4) Plasma Treatment

106 Chitosan-based films with incorporated CNCs were cut out in different
107 sizes, according to the further analysis. One sample at the time was placed
108 on the microscopic glass and fixed with carbon tape. The sample was then
109 introduced into the 80 cm long discharge tube made of borosilicate glass
110 with diameter 4 cm. The tube was pumped with a two-stage rotary pump

111 of a nominal pumping speed of $80 \text{ m}^3 \text{ h}^{-1}$. The base pressure was 1 Pa.
112 Plasma was sustained by a coil with six turns which was connected to the RF
113 generator via a matching network. The generator operated at the standard
114 frequency of 13.56 MHz. The output power was set to 150 W, which was
115 found to be the optimal among preliminary tested 150 W, 200 W and 400 W
116 (data not shown). At these conditions, diffusing plasma expanded far away
117 from the coil. The sample was positioned 7 cm away from the coil. The
118 treatment was carried out with CF_4 gas at 50 Pa for 0.5 s, 1 s, 3 s, 5 s, 10 s,
119 20 s and 30 s.

120 *2.4. Films Surface Analyses*

121 Water contact angle (WCA), analyzed with Drop Shape Analyser DSA-
122 100 (Krüss GmbH, Hamburg, Germany), was measured at two points in this
123 study: immediately after the treatment on all of the samples and through
124 period of 40 days on the selected sample (treated for 5 s). A static contact
125 angle was determined using a sessile drop method. MilliQ water with the
126 volume of the drop of 1 μL was used for the measurements. The first set
127 of measurements was applied to select the optimal treatment time and the
128 second to follow the effect of aging onto the newly reached hydrophobic char-
129 acter of the surface. The surface chemical composition of the samples was
130 analyzed by means of X-ray Photoelectron Spectroscopy (XPS) using instru-
131 ment TFA XPS (Physical Electronics, Munich, Germany). Monochromatic
132 Al $\text{K}\alpha_{1,2}$ radiation at 1486.6 eV over an area of $400 \mu\text{m}^2$ was used as a source

133 for excitement. Hemispherical analyzer positioned at an angle of 45° with
134 respect to the sample surface was applied for the detection of photoelectrons.
135 The survey spectra were measured at a pass energy of 187 eV with an energy
136 step of 0.4 eV. High-resolution XPS spectra of carbon C1s were measured at
137 a pass energy of 29.35 eV with an energy step of 0.125 eV. For surface charge
138 neutralization an additional electron source was used. The C-C component
139 in C1s was set to the binding energy of 284.8 eV. Analysis of the obtained
140 spectra was carried out using MultiPak v8.1c software (Physical Electronics,
141 Munich, Germany). The C1s spectra were fitted with the Gauss-Lorentzian
142 function, where the width and positions of the peaks were fixed during the
143 fitting procedure. FTIR spectra of the untreated samples, treated samples
144 right after the treatment and after 31 days were recorded with ATR-FTIR
145 Spectrum Two (Perkin Elmer, Germany) from 4000 cm^{-1} to 400 cm^{-1} with
146 4 cm^{-1} step, accumulation of 64 scans. All measurements were done in par-
147 allels. Treated and untreated samples were coated with 10 nm layer of gold
148 and were subjected to SEM analysis with Supra 35VP electron microscope
149 (Carl Zeiss, Jena, Germany).

150 *2.5. Film Physico-Chemical Properties*

151 Film thickness and mechanical properties of the film samples were deter-
152 mined according to Bajić et al. (2020), with slight modification in the latter
153 (size of the samples was 6 x 2 cm and gauge length segment was 4 cm). For
154 a detailed description of the analysis the reader is referred to Appendix A.

155 Protocol described by [5] was followed for assessment of water vapor trans-
156 mission (WVT). Brief description can also be found in Appendix A. During
157 the aging process the samples were stored at 50 % RH and room temperature.

158 *2.6. Film stability in liquid environments*

159 To observe stability of fluorinated compounds in liquid environments, pro-
160 cessed sample was cut into smaller pieces with mass approximately 10 mg and
161 submerged into liquid media (water, 5 % acetic acid_(aq), 10 % ethanol_(aq)) of
162 appropriate volume so the final concentration was 2 mg_{film} mL⁻¹. The sam-
163 ples were collected after 144 h (this time was selected to achieve the longest
164 possible submersion time while avoiding disintegration of the sample, visible
165 after 168 h), dried under nitrogen flow and the solid residues were redis-
166 solved in 70 % acetonitrile and analyzed with LC-ESI-MS system in positive
167 and negative ionization mode. UHPLC-MS system (Accela 1250, coupled to
168 an LTQ Velos MS, Thermo Fisher Scientific, Waltham, MA, USA) was used
169 to analyze liquid media upon contact with reference and CF₄ treated films).
170 HILIC LC-MS analyses in ESI(+) and ESI(-) mode were carried out on Phe-
171 nomenex Luna NH₂ (100 Å, 100 × 2.0 mm i.d., 3 µm) column using 0.05 %
172 ammonia in water solution (mobile phase A) and acetonitrile (mobile phase
173 B) with gradient elution from 10 % to 40 % A in 10 min at a flow rate of 300
174 µL min⁻¹. Re-equilibration of the column with initial conditions was applied
175 from 11th to 20th min before each next injection. Column oven and autosam-
176 pler temperature were set at 25 °C and 5 °C, respectively. Injection volume

177 was 2 μ L. Alternative HILIC and RP methods are described in Appendix A.
178 MS parameters were optimized using glucosamine standard (25 μ g/mL) and
179 were as follows: heater temperature 300 °C, sheath gas 41 a.u., auxiliary gas
180 35 a.u., sweep gas to 0 a.u., spray current 5 μ A, capillary temperature 200
181 °C and S-Lens RF Level 49%. The MS spectra were acquired in the m/z
182 range of 50–2000. The collected chromatograms and spectra were evaluated
183 using the Xcalibur software (version 2.1.0, Thermo Fisher Scientific).

184 **3. Results and Discussion**

185 *3.1. Film Surface Analyses*

186 The produced chitosan/cellulose nanocrystals films subjected to CF₄ plasma
187 for various time duration (0.5 s, 1 s, 2 s, 5 s, 10 s, 20 s and 30 s). Water
188 contact angle (WCA), that was evaluated immediately after the treatment,
189 increased abruptly from 94 \pm 4 ° to 121 \pm 2° in the first two seconds and
190 then stabilized between 122 ° and 125 ° with further processing as shown in
191 Figure 1a. WCA reached with treatment is higher than water repelling of bio-
192 polymer films (75 ° for chitosan-only, 58 ° for alginate reinforced with CNCs,
193 23 ° for cellulose nanofibrils films and around 52 ° for starch film) [14, 43], as
194 well as conventional polymers (PE 102 °, PS 91 °, PVC 87 ° and PET 81 °)
195 [44]. To reveal the cause of this occurrence, the samples were inspected with
196 XPS that showed the saturation of the surface with fluorine in the first two
197 seconds of processing, which correlates to the increase of WCA and its sta-
198 bilization (Figure1b). The initial C:O:N ratio (69:27:4), was notably altered

199 upon fluorination. The composition at saturation with fluorine, that is after
 200 two seconds of processing, was 41 % carbon, 48 % fluorine, 10 % oxygen and
 201 1 % nitrogen. It was also observed that the ratios slightly change, namely
 202 higher carbon, oxygen and nitrogen content and lower fluorine percentage,
 203 when plasma is operating longer, which could be attributed to the etching
 204 of the surface. Based on these observations, 5 s was chosen as an optimal
 205 treatment time and was used for further analyses.

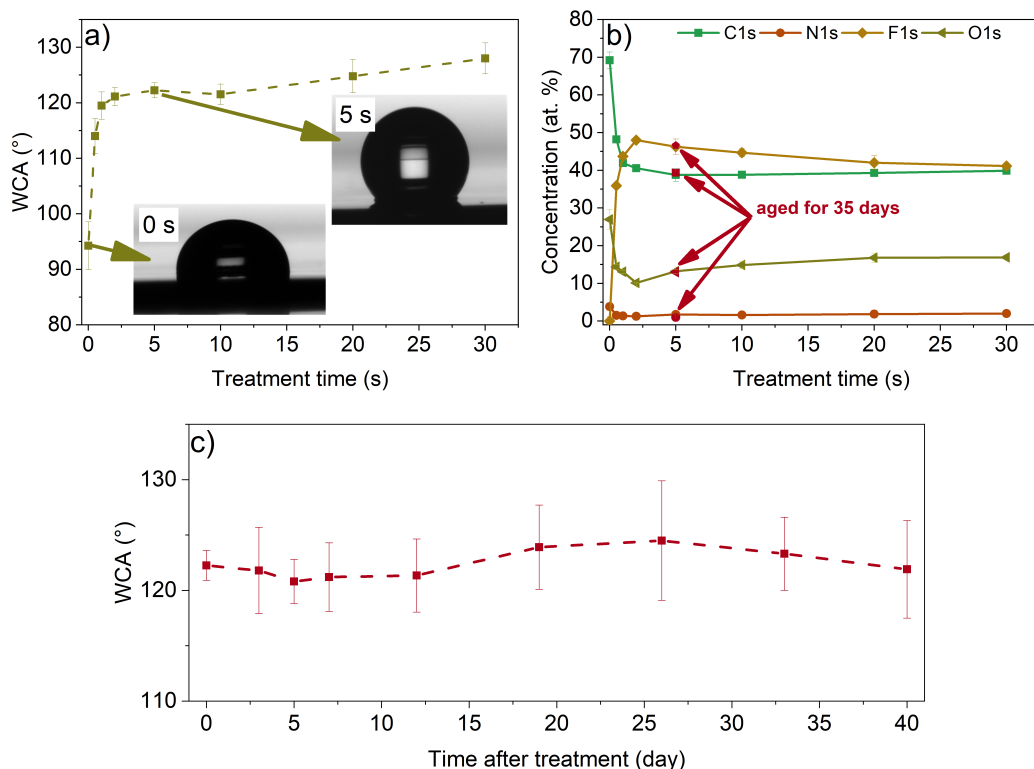


Figure 1: a) WCA of the samples with respect to treatment time, b) atomic composition at various treatment times and atomic composition of the sample treated for 5 s after 35 days (marked in red) and c) stability of WCA over a period of 40 days.

206 The treated films remained semi-transparent, slightly brown in color and

207 slightly sticky, indicating that plasma treatment had no effect detected with
208 a naked eye. However, plasma treatment can cause etching on the surface of
209 the film, surface morphology was closely inspected with SEM before and after
210 processing. The micrographs shown in Figure 2 are taken at magnifications
211 1 k, 10 k and 20 k from left to right. At the smallest magnification, the
212 reference and the treated surface seem smooth with no visible roughness or
213 porosity. Further magnification, however, reveals the structure of rod-like
214 shaped CNCs in the film. The particles seem to be evenly distributed in
215 the chitosan matrix. The surface retained the same morphological properties
216 after the modification.

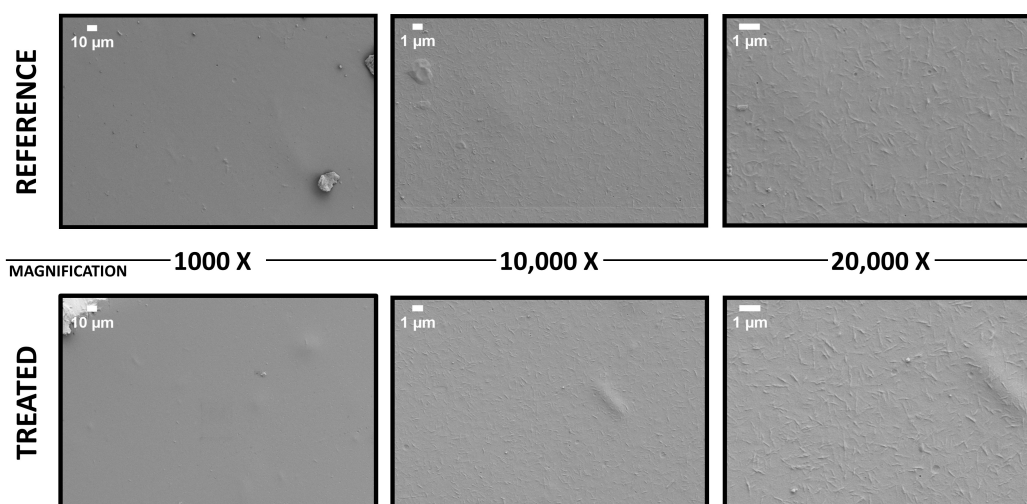


Figure 2: SEM images of reference and treated sample at three different magnifications.

217 Alternative to conventional polymers used for packaging applications is
218 required to exhibit stable composition and water repelling character over
219 longer period of time. Observations based on the following of WCA over the

220 period of 40 days showed that it remained constant at around 120 °. Addi-
221 tionally, the composition was shown to remain the same after 35 days of aging
222 with C:F:O:N ratio 39:47:13:1. Hereby, the interest in the surface chemistry
223 throughout the plasma treatment arose, so high-resolution XPS carbon spec-
224 tra of samples processed for various times were recorded (Figure 3a). To gain
225 a deeper insight into C-F related binding contributions the deconvolution of
226 the C 1s peaks was performed for the reference sample and sample treated
227 for 5 s. In the untreated sample, the C-C (284.8 eV), C-O (286.5 eV), O-C-O
228 (288.0 eV) and O-C=O (289.0 eV) bonds were detected. The intensity of the
229 peak corresponding to initially prevalent C-C bond decreases with plasma
230 processing, while at the same time, peaks related to various C-F bindings
231 start to increase. With longer treatment times the peaks at higher binding
232 energies, that correspond to carbon atoms binding to more than one fluorine,
233 become more prominent. It can be observed that the intensity of the peak at
234 293.5 eV, corresponding to CF₃ bond, is increasing in the first five seconds
235 of the treatment but this trend takes a turn and the peak starts to decrease
236 with further processing. On the other hand, at the same time, the peak
237 featuring the C-C bond starts to increase after initial decrease in the first
238 10 s of the treatment that, again, could be due to the etching of the already
239 treated surface and is in alignment with the previously described alteration
240 of atomic ratios. Further, the high-resolution C 1s XPS spectrum of sample
241 treated for 5 s was inspected into detail (Figure 3c). The shape of the treated
242 sample's spectrum differentiates greatly compared to the reference because

243 of the formation of various fluorine groups. Decrease in the intensity of the
244 initially present peaks corresponding to C-C, C-O and O-C=O bonding was
245 detected, with the latter completely disappearing upon treatment, while the
246 peak positioned at 288.0 eV met a slight increase. However, this same peak
247 can be associated with CF-CF binding as well which is most possibly respon-
248 sible for the inflation. The treated sample is featured by the newly developed
249 peaks attributed to CF-CF₂ (289.5 eV), CF₂-CF₂ and CF-CF₂ (291.5 eV),
250 CF₃ (293.4 eV) and O-CF₂, O-CF₃ (295.0 eV).

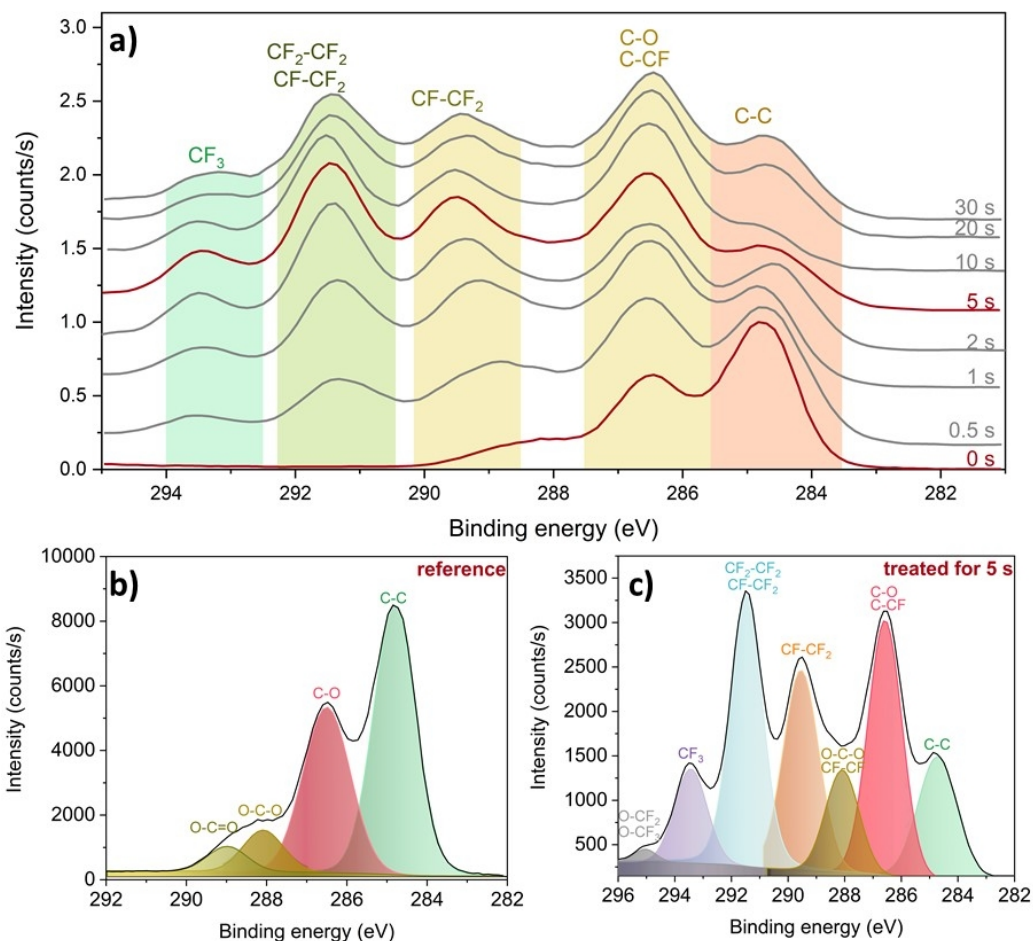


Figure 3: a) Evolution of high resolution XPS carbon spectra according to plasma processing time; b) and c) deconvolution of high-resolution C 1s XPS spectrum of reference material and sample treated for 5 s, respectively.

251 To further understand possible structural changes upon fluorination, the
 252 sample treated for 5 s were additionally analyzed by means of FTIR-ATR and
 253 compared to the reference spectrum. Both spectra exhibited characteristic
 254 peaks for chitosan namely: the broad peak between 3500 cm^{-1} and 3030
 255 cm^{-1} (corresponds to O-H and N-H bands in chitosan), the two weak bands

256 located between 2820 cm^{-1} and 3015 cm^{-1} (attributed to C-H stretching in
257 alkane groups), peaks appearing between 1490 cm^{-1} and 1750 cm^{-1} (assigned
258 to C=O stretching - carbonyl and amide I, and N-H bending - amide II,
259 respectively) [45, 46]. The shapes of the treated and non-treated spectra
260 differentiate in the four different regions marked I-IV in Figure 4. The peaks
261 in region positioned between 2830 cm^{-1} and 3000 cm^{-1} , that are attributed
262 to alkane C-H stretching, the absorbance decreases upon plasma processing
263 pointing to lower abundance of these groups. Similar trend is visible in
264 regions II and III, both corresponding to C-O stretching, where the peak
265 positioned at 1287 cm^{-1} disappears and the peaks between 1130 cm^{-1} and
266 1060 cm^{-1} straighten after fluorine treatment. Similarly, examining region
267 IV, corresponding to C-H bending, it can be observed that the intensity
268 decreases.

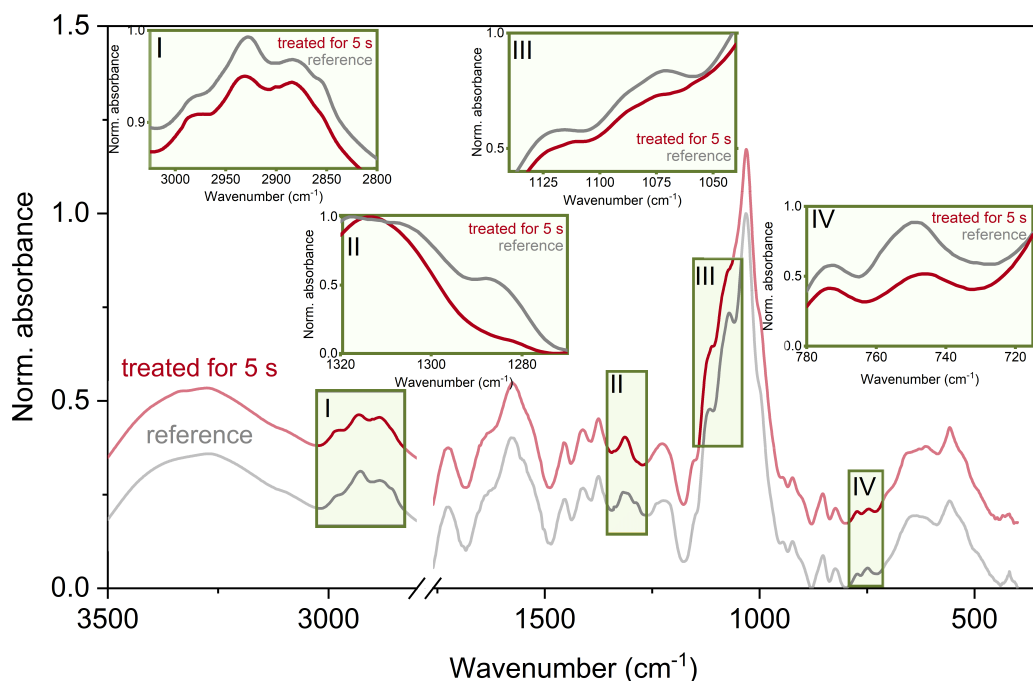


Figure 4: FTIR-ATR spectra of reference and sample treated for 5 s immediately after processing with marked regions where the structural changes were observed.

269 3.2. Physico-chemical Properties

270 For packaging important properties, namely tensile strength (TS), elongation at break (ϵ) and water vapor transmission (WVT) were measured
 271 immediately after the treatment as well as after 31 days of aging in the controlled environment. As seen in Figure 5, TS and ϵ did not decline with
 272 hydrophobization, which was of concern with this treatment. On the contrary, TS improved by 68% (from initial 1.38 ± 0.37 MPa to 2.32 ± 0.51
 273 MPa after the treatment). Similarly, the ϵ increased from the original $63 \pm 13\%$ to $78 \pm 10\%$. It is worth noting, that both of the measured me-
 274 chanical properties are strongly influenced by the thickness of the film and
 275
 276
 277
 278

279 moisture content (MC). While the thickness of the films (0.10 ± 0.01 mm)
280 did not deviate upon modification, MC decreased in the fluorinated samples
281 from initial $11.6 \pm 0.4\%$ to $9.2 \pm 0.3\%$. The water is bound in the sam-
282 ple through hydrogen bonding sites in chitosan (Bajić et al., 2020), so the
283 lower amount of moisture after treatment might be attributed to saturation
284 of hydrogen binding sites of chitosan and CNCs with fluorine. As moisture
285 has a plasticizing effect on the film, TS and MC are in inverse correlation
286 [47], which explains the increase in TS of the treated sample well. However,
287 generally, elongation rate has positive correlation to MC, which is not the
288 case in the fluorinated samples. The newly formed bonds between fluorine
289 and chitosan-CNCs might contribute to higher stiffness of the material. The
290 mechanical properties of biopolymer films strongly relate to free and bound
291 water [7]. Leceta et al. suggest that the water bound inside the crosslinked
292 composite matrix tends to form hydrogen bonds with more polar functional
293 groups, which are in this case the newly added fluorine-related groups on the
294 surface. With this, the internal film structure is damaged, resulting in loss
295 of strength, stiffness and lower stretching [48]. Furthermore, high resolution
296 spectrum of the aged sample was thoroughly inspected (Fig. A2), suggesting
297 that the bonds, disrupted during the aging process, are in the backbone of
298 the polymer chains, which could negatively affect the mechanical properties
299 shown in Figure 5. Aging of the films leads to loss of the advantage in me-
300 chanical properties gained by treatment. MC in the reference sample remains
301 constant, whereas in the fluorinated one the moisture content increased to

302 $13 \pm 1.0\%$ and consequently a drop in TS was seen. Based on assumption
303 that the moisture content was lower in the treated film because of formation
304 of hydrogen bonds between chitosan-CNCs and fluorine, the increase over
305 time could be attributed to substitution of these binding with water-chitosan
306 again. Both reference and processed films age into a stiffer material with ϵ 59
307 $\pm 14\%$ and $56 \pm 13\%$, respectively. Important role of packaging is to shelter
308 the contents from loss of moisture or to prevent the moisture from the envi-
309 ronment to breach into the inside and effect the contents. Upon treatment,
310 water vapor transmission (WVT) lowered (from initial $51.3 \pm 0.4 \text{ g cm}^{-2}$
311 day^{-1} to $36.5 \pm 3.0 \text{ g cm}^{-2} \text{ day}^{-1}$) as shown in Figure 5, which could be at-
312 tributed to increase in hydrophobic character. Comparing to other available
313 materials, the processed sample is better at preventing water transmission
314 than PVA, nanocellulose-based films, cellophane and PCL and comparable
315 to PS, Nylon 6 and PLA [16]. With aging, WVT slightly decreases in both
316 samples, with treated sample still performing better than the reference.

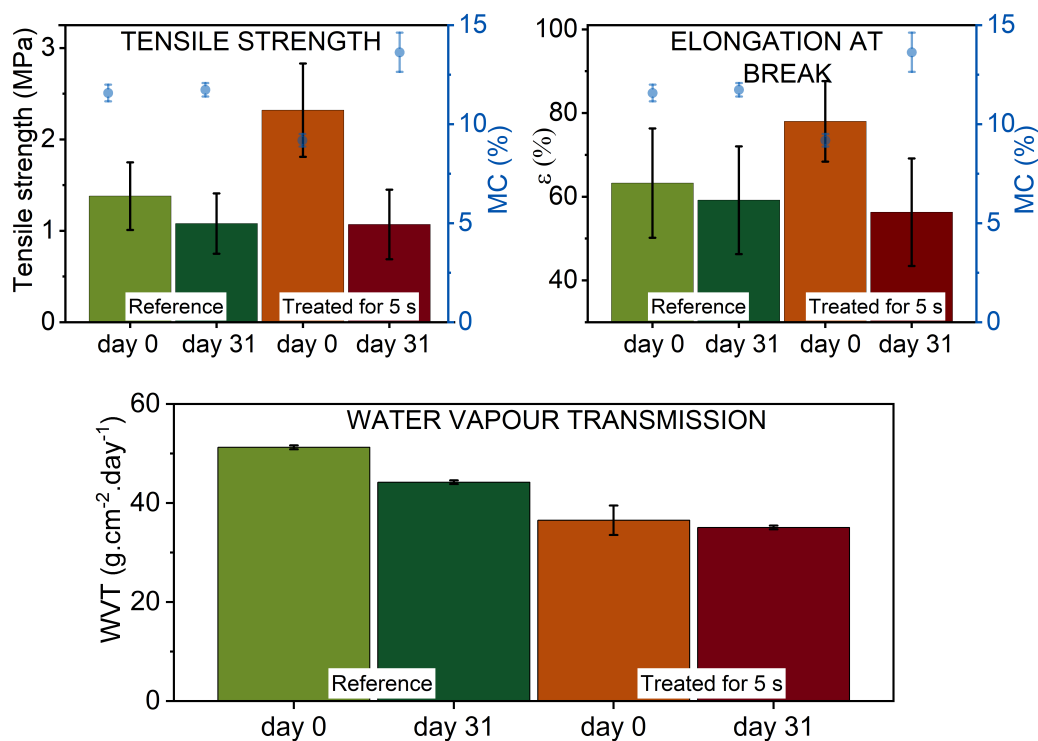


Figure 5: TS, ϵ and WVT of the reference and sample treated for 5 s immediately after processing and after aging in controlled environment for 31 days.

317 3.3. Film stability in liquid environments

318 To further demonstrate suitability of the plasma treated material in pack-
 319 aging industry, which was suggested by the measured physicochemical prop-
 320 erties and long-term stable hydrophobicity, the interaction of the films with
 321 liquid environments was evaluated. The main concern related to material'
 322 s adequacy as packaging is potential leaching of fluorinated compounds into
 323 the environment, which was explored with an untargeted analysis of film
 324 leachates in water, 5 % acetic acid_{aq}, 10 % ethanol_{aq} in both HILIC and RP
 325 modes using ESI(+)-MS and ESI(-)-MS. Two HILIC methods and an RP

326 method were used to analyze the samples. LC-MS chromatograms of treated
327 samples were compared with those of reference (blank) samples to reveal the
328 potential leaching of fluorinated compounds. Both HILIC methods showed
329 comparable results. In RP method, all peaks eluded in dead time (as ex-
330 pected for polar compounds such as glucosamine and N-acetyl glucosamine,
331 potentially present in the simulants as monomer units of chitosan). The
332 experiments showed no difference between the reference and treated sam-
333 ple, except for the water samples (LC-ESI(+)-MS chromatograms and MS
334 spectra shown in Figure 6; supplementary LC-MS chromatograms of the
335 leachates are presented in Fig. A2. In the reference sample, the peak at
336 t_R 6.8 min (marked with I) corresponds to the signal at m/z 586, which
337 matches a formula of a trimer consisting of one glucosamine (GlcN) unit and
338 two *N*-acetyl-D-glucosamine (GlcNAc) units. LC-MS analysis of treated wa-
339 ter sample showed three split peaks, for which the MS analysis revealed that
340 both parts correspond to the same m/z , so the splitting could be attributed
341 to the existence of isomerism or aminosugar anomerism. The peaks with
342 t_{RS} between 4.2-5.0 min (peaks III and IV) correlate to m/z 180, which is
343 attributed to protonated GlcN. The following two peaks with t_{RS} between
344 5.6-6.0 min (peaks V and VI) and between 6.6-7.2 min (peaks VII and II)
345 correspond to m/z at 341 (protonated GlcN/GlcN dimer) and m/z at 544
346 (protonated GlcN/GlcN/GlcNAc trimer), respectively [49]. Additionally, the
347 peak with t_R 6.8 min in the treated sample (peak II) that coincide with the
348 t_R of peak I in the control sample showed additional mass peak of lower

349 intensity (m/z at 586), which was attributed to GlcN/GlcNAc/GlcNAc [49].
350 There were no detected peaks pointing to traces of fluorinated compounds
351 in both HILIC and RP using either positive or negative ionization mode,
352 however some limitations apply to this approach. The MS parameters were
353 tuned using glucosamine as a standard due to the non-commercially available
354 fluorinated standards. Moreover, taking into consideration that only surface
355 was modified, it can be assumed that the concentration of potential fluori-
356 nated compounds leached into the liquid media is rather low, and may be
357 under the limit of detection (LOD) of our methods, however LODs cannot
358 be calculated due to the absence of standards.

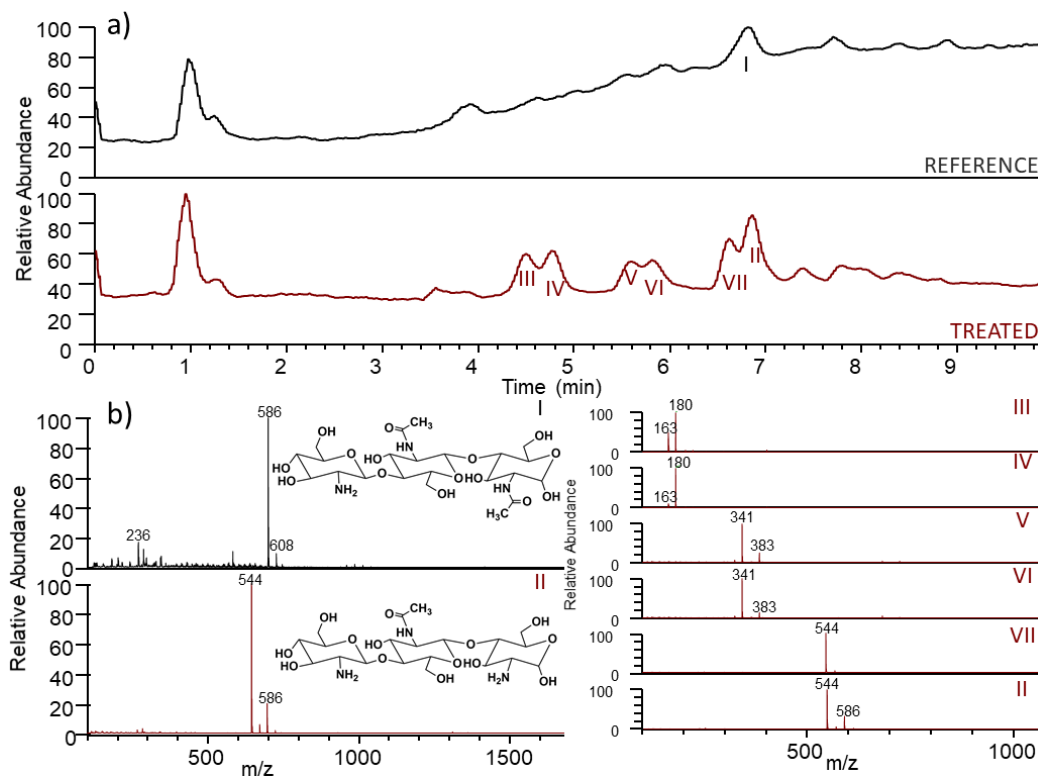


Figure 6: HILIC LC-ESI(+)-MS chromatograms of water leachates in which reference (black) and sample treated for 5 s (red) were submerged for 144 h (a) with marked peaks further analyzed with MS and their corresponding spectra (b).

359 4. Conclusions

360 Hydrophobization of bioplastic bio-composite formed from chitosan with
 361 incorporated cellulose nanocrystals have been achieved using ultrafast and
 362 efficient RF-generated plasma treatment. The main effect of the hydropho-
 363 bization was to improve water-related properties and raise the material's
 364 suitability for packaging applications. After only 5 s of treatment, the in-
 365 crease of WCA by 28° was achieved. The insight into the surface chem-
 366 istry behind these changes was provided, with XPS revealing newly formed

367 fluorine-related bonds, while a decrease in C–H and C–O stretching and
368 C-H bending was detected with FTIR-ATR. It is worth noting, that the
369 surface was not damaged by the treatment. Plasma processing decreased
370 MC in the fabricated films, which influenced TS, ϵ and WVT. However, for
371 the role of packaging, the modified sample performed better than the ref-
372 erence one. Stability of the hydrophobic coating, an important aspect for
373 this application, was demonstrated by following WCA over 40 days as well
374 as determining comparable atomic composition with XPS after 35 days of
375 aging. On the other hand, TS, ϵ and WVT all decreased in both, treated
376 and reference sample with time, causing the processed material to lose its
377 superiority. Lastly, to investigate the material's interaction with liquid en-
378 vironments and potential leaching of fluorinated compounds, the films were
379 immersed in three different liquid environments (water, 5% acetic acid, 10%
380 ethanol) and analyzed by means of LC-MS. No evidence pointing to presence
381 of fluorinated compounds were found.

382 **Author Contributions**

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387 tion, Data Curation, Writing - Review & Editing. **Blaž Likozar:** Writing -
388 Review & Editing, Supervision, Project administration, Funding acquisition,

389 **Uroš Novak:** Conceptualization, Writing - Review & Editing, Supervision,
390 Project administration, Funding acquisition

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